

*Application of thermal-stress models to  
ideal taper, maximum casting speed to  
avoid breakouts, and the prediction of  
strand width variations*

*Chunsheng LI*

*Department of mechanical & industrial engineering  
University of Illinois at Urbana-Champaign*

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# *Acknowledgement*

- *Continuous Casting Consortium (CCC) members*
- *National Center of Supercomputing Applications*
- *Professor Brian G. Thomas*
- *Ya Meng*

# *Outline*

- *Fixed-Grid finite-element model of mechanical behavior of solidifying metals(CON2D)*
  - *Model Description*
  - *Validation*
- *CON2D Applications:*
  - *Ideal Taper Prediction*
  - *Maximum Casting Speed to Avoid Breakouts*
  - *Prediction of Strand Width Variations*
- *Future Work*

***Fixed-Grid finite-element model of  
mechanical behavior of solidifying  
metals (CON2D)***

# *Model Description*

- *Finite element thermal stress model*
- *Phase fractions from non-equilibrium Fe-C phase diagram for plain carbon steel*
- *Recalescence and kinetics neglected*
- *Linear phase fraction model between liquidus and solidus for ferritic and austenitic stainless steels*
- *2-D generalized plane strain*

$$\dot{\mathbf{e}}_{total} = \dot{\mathbf{e}}_{elastic} + \dot{\mathbf{e}}_{plastic/creep} + \dot{\mathbf{e}}_{thermal} + \dot{\mathbf{e}}_{flow}$$

# *Model constitutive equations*

- *Mizukami elastic modulus data*
- *Kozlowski constitutional equations for austenite, and modified model for delta-ferrite:*
  - *Kozlowski Model for Austenite*

$$\dot{\mathbf{e}} = f(c) \left[ \mathbf{s} - f_1(T) |\mathbf{e}|^{f_2(T)} \text{sgn}(\mathbf{e}) \right]^{f_3(T)} \exp\left(-\frac{A}{T}\right)$$

$$\text{where : } f_1(T) = 130.5 - 5.128 \times 10^{-3} T$$

$$f_2(T) = -0.6289 + 1.114 \times 10^{-3} T$$

$$f_3(T) = 8.132 - 1.54 \times 10^{-3} T$$

$$f(c) = 4.655 \times 10^4 + 7.14 \times 10^4 c + 1.2 \times 10^5 c^2$$

# *Model constitutive equations(Continued)*

– *Modified Power Law Model for d-ferrite*

$$\dot{\epsilon} = 0.1 \left| \frac{\mathbf{S}}{f(c) \left( \frac{T}{300} \right)^{-5.52} (1 + 1000\epsilon)^m} \right|^n$$

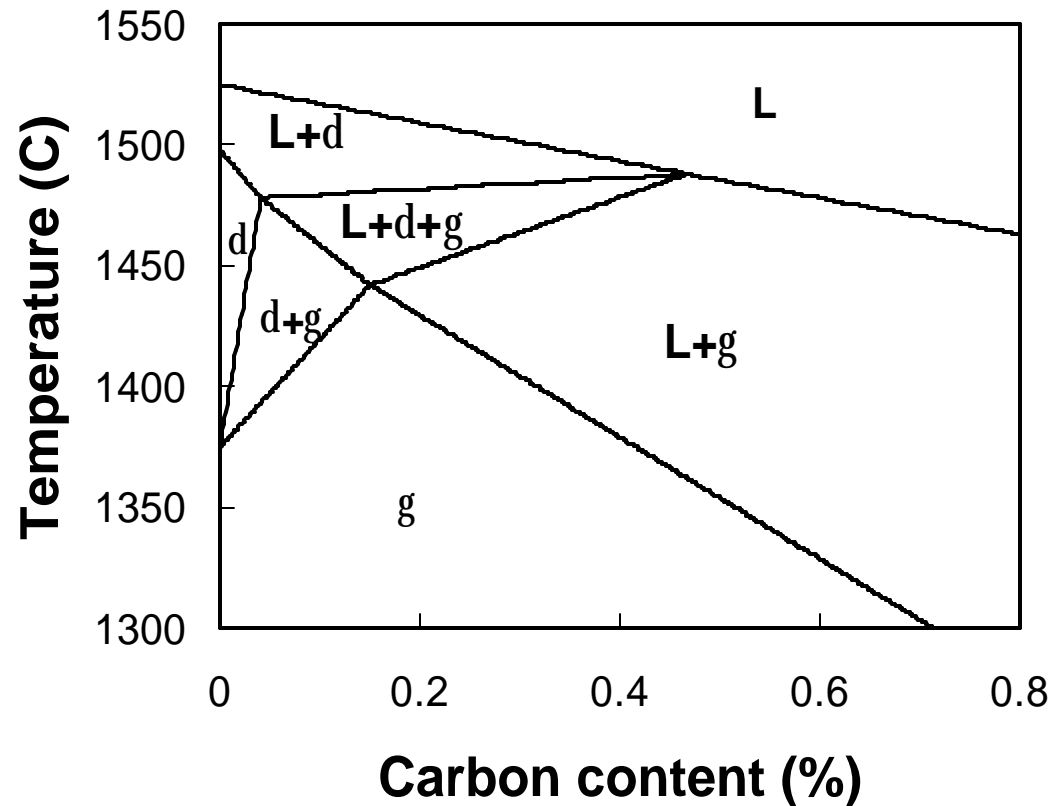
*where :*

$$f(c) = 1.3678 \times 10^4 c^{-5.56 \times 10^{-2}}$$

$$m = -9.4156 \times 10^{-5} T + 0.3495$$

$$n = \frac{1}{1.617 \times 10^{-4} T - 0.06166}$$

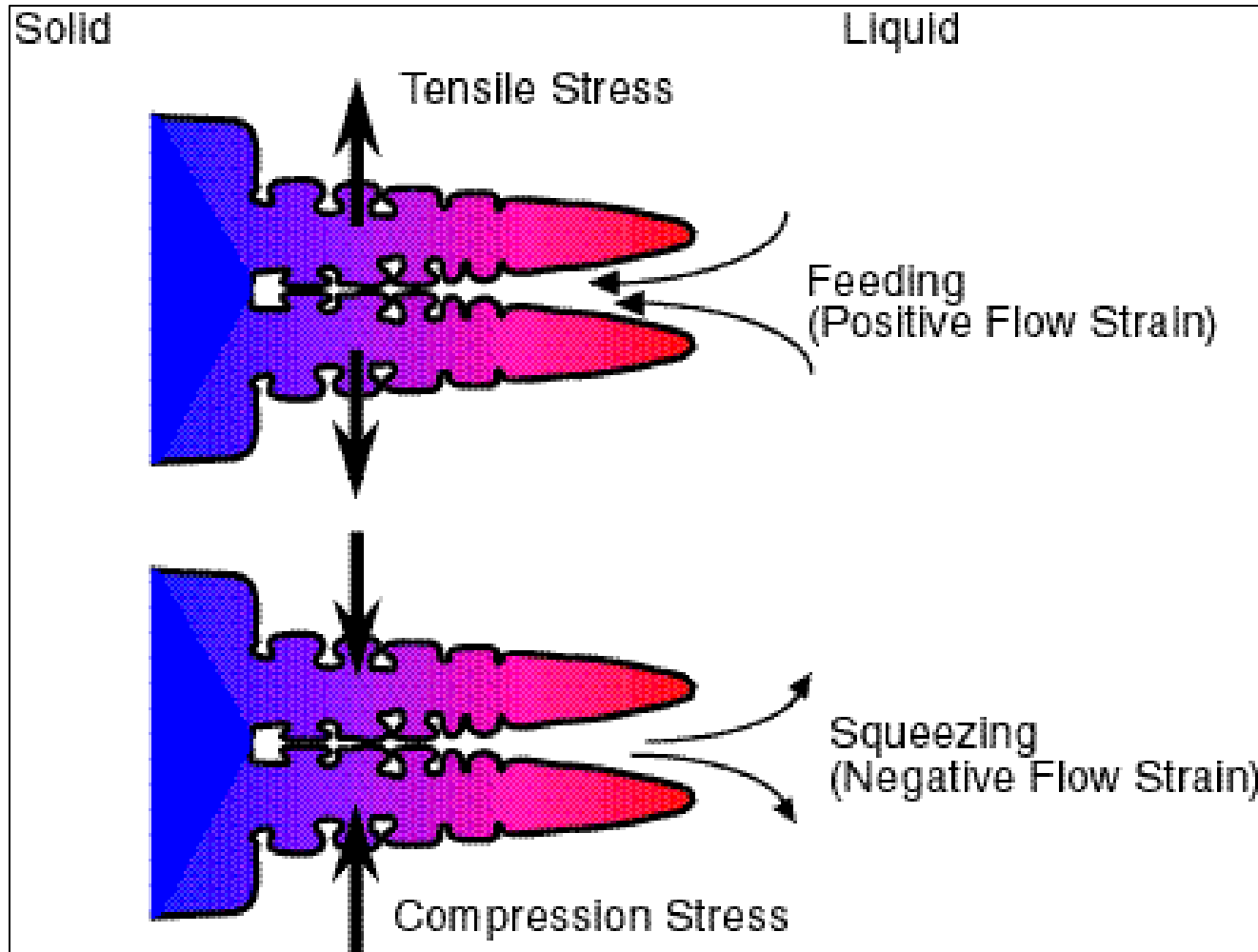
# *Non-equilibrium phase diagram\* of plain carbon steels\*\* used in CON2D*



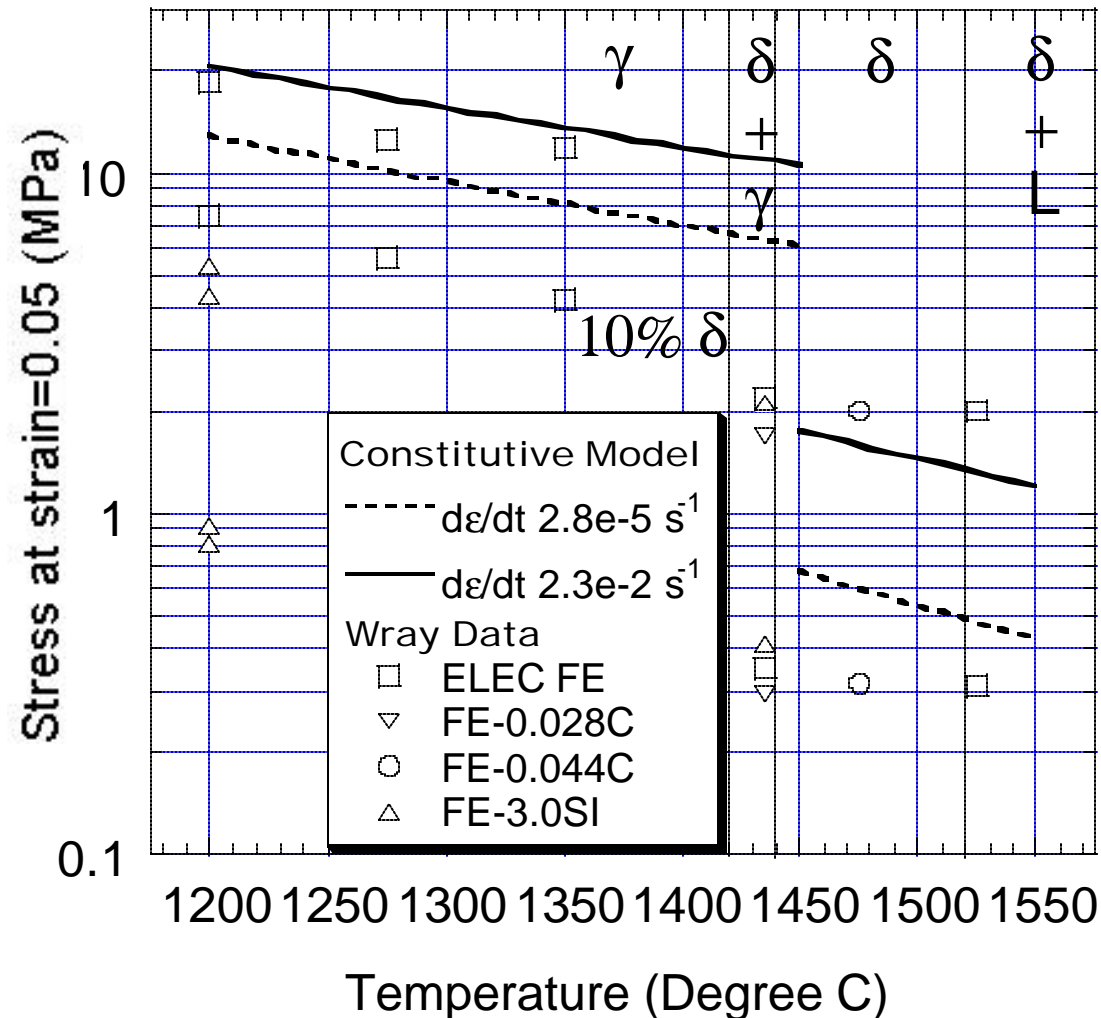
- *\*Young Mok WON et. al., Effect of Cooling Rate on ZST, LIT, ZDT of Carbon Steels Near Melting Point”, ISIJ International, Vol. 38, 1998, No. 10, pp. 1093 –1099*
- *\*\*Other Steel Components: 1.52%Mn, 0.34%Si, 0.015%S, 0.012%P*



# *Flow Strain Concept*

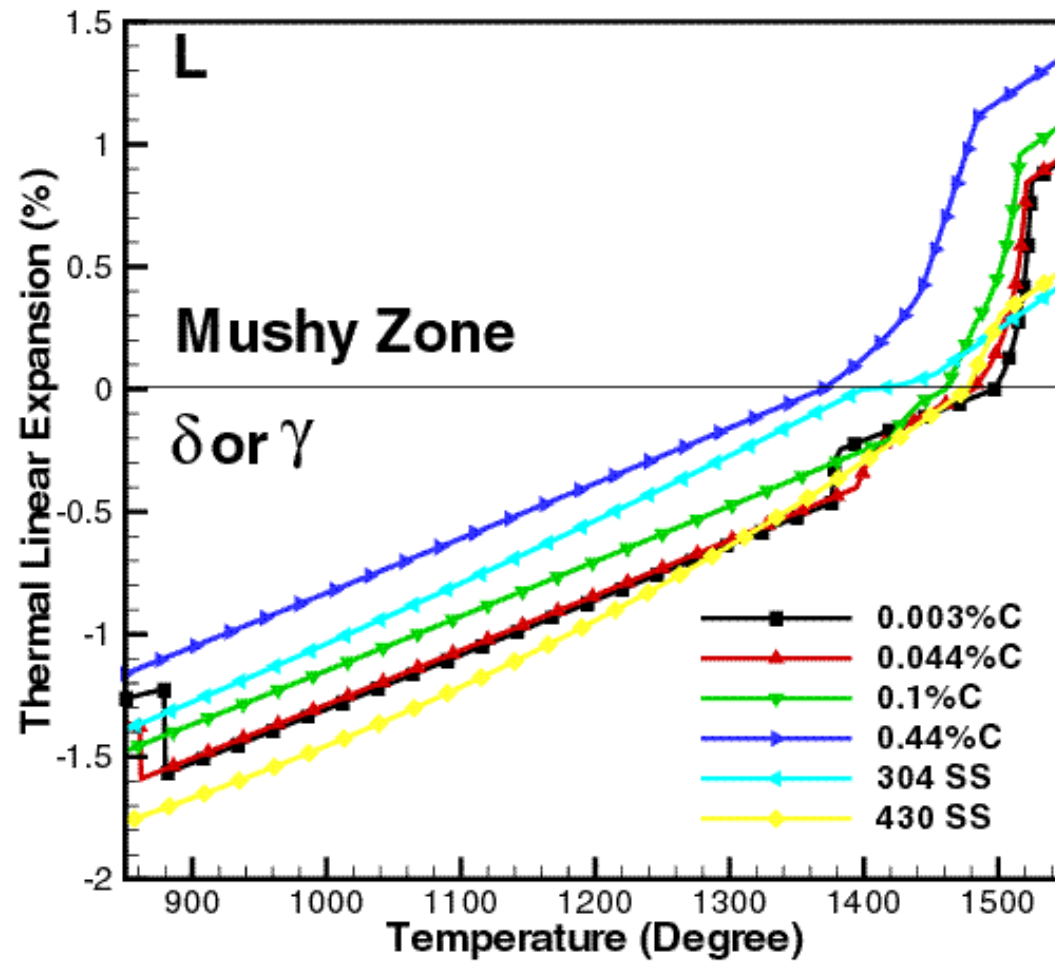


# Creep Model Validation\*



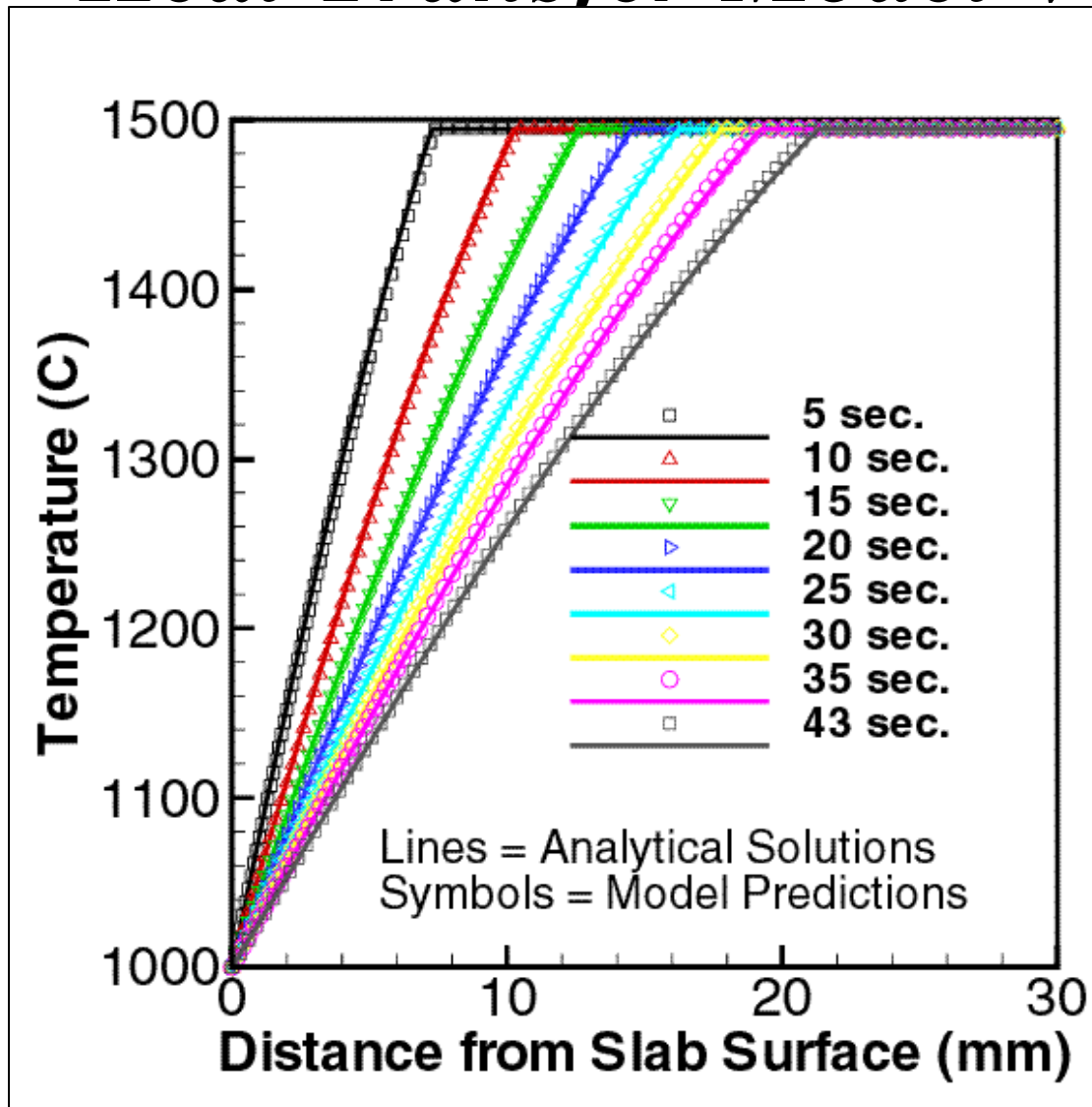
\* From B. G. Thomas and J. T. Parkman, *Simulation of Thermal Mechanical Behaviour During Initial Solidification*, Thermec '97 International Conference on Thermomechanical Processing of Steel and Other Materials, Wollongong, Australia, 1997

# *Thermal Linear Expansion*



- Liquid data from: Jimbo & Cramb, Met. Trans. B, 24B, 1993, 5-10
- Solid data for plain carbon steel from: Harste, Jablonka & Schwerdtfeger, 4<sup>th</sup> Int. Conf. On Continuous Casting, CRM, 1988, Brussels, 633-644
- 304 stainless steel data from: Thermophysical Properties of Materials. Curve 28-32, pp1151-1152.
- 430 stainless steel data from: Thermophysical Properties of Materials. Curve 52, pp1151-1152.

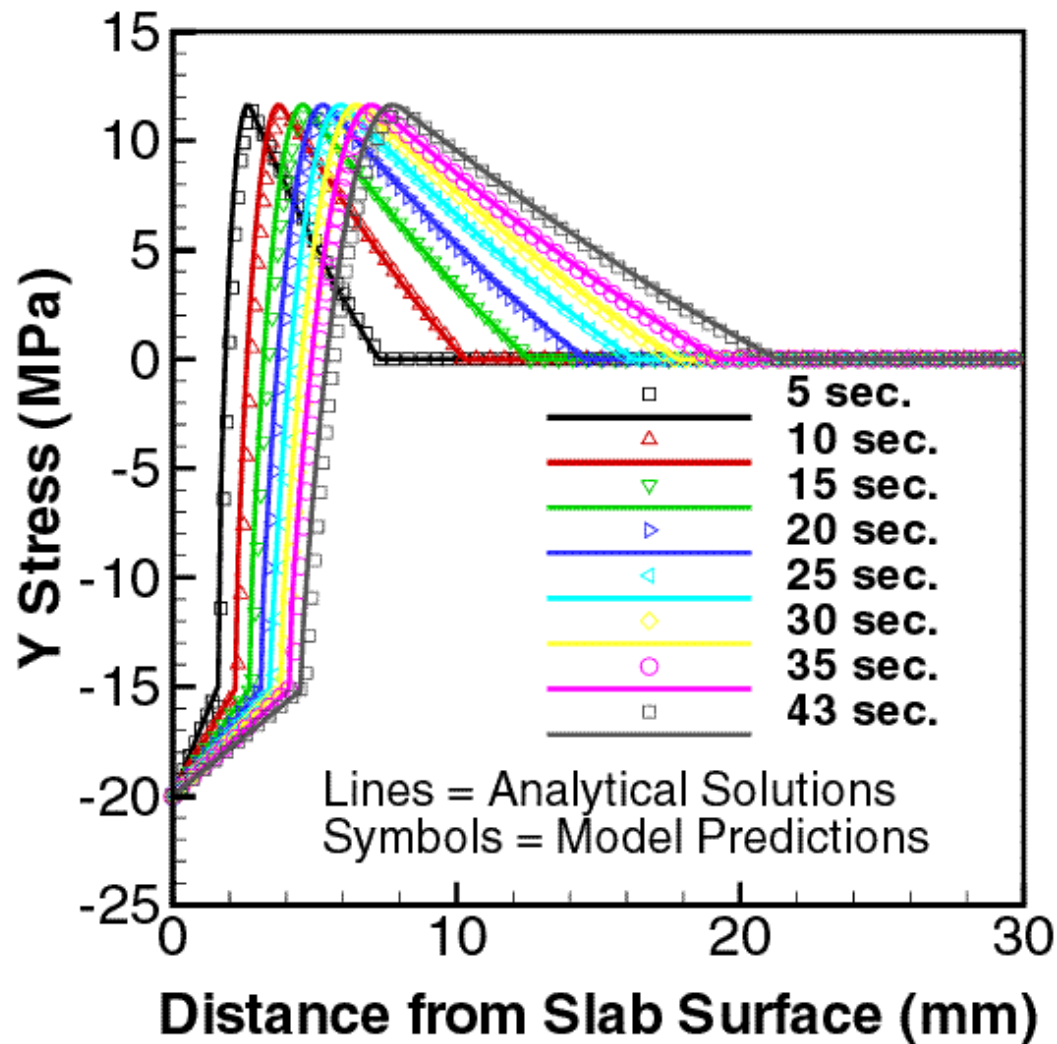
# Heat Transfer Model Validation



- *Lines: Boley & Weiner's analytical solution\**
- *Dots: CON2D computation results*

\* *J. H. Weiner and B. A. Boley, J. Mech. Phys. Solids, 1963, Vol. 11, pp145-154*

# *Stress Model Validation*

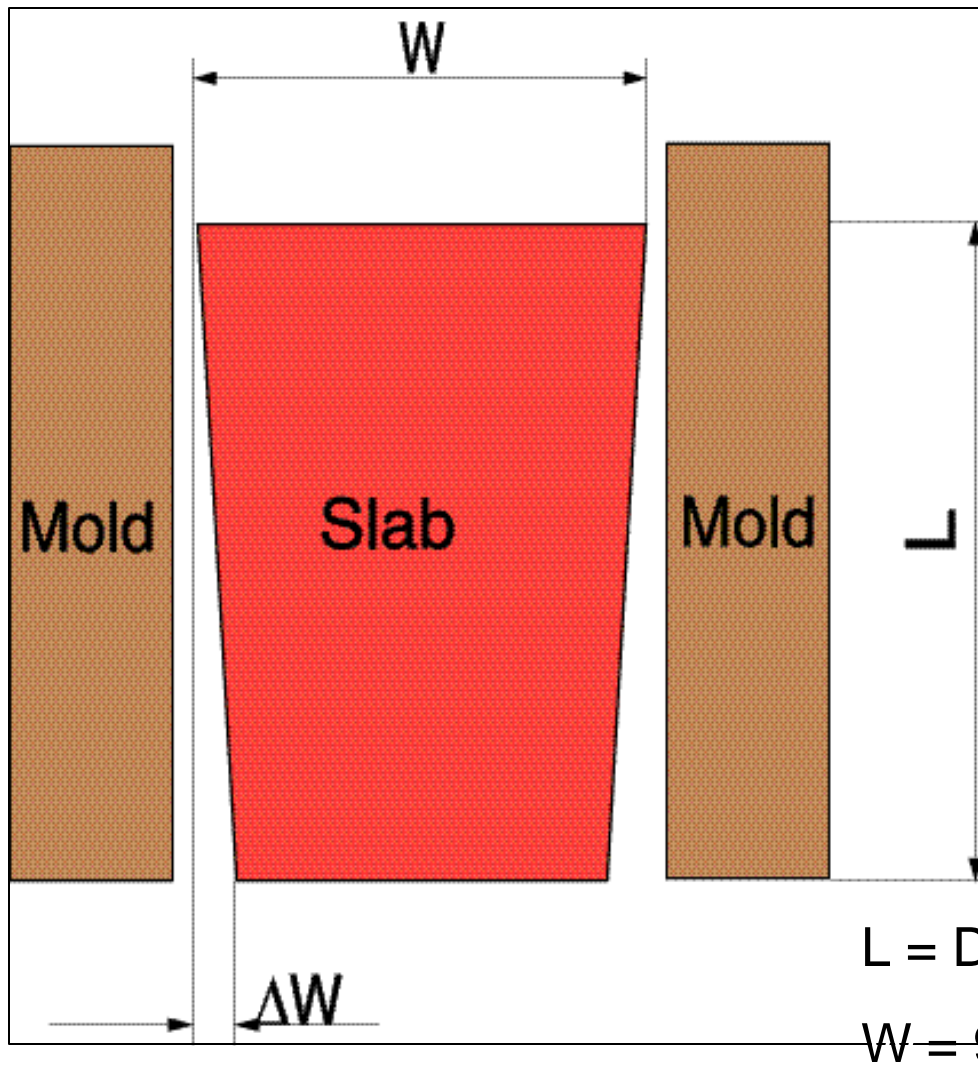


- *Lines: Boley & Weiner's analytical solution\**
- *Dots: CON2D computation results*

\* *J. H. Weiner and B. A. Boley, J. Mech. Phys. Solids, 1963, Vol. 11, pp145-154*

# ***Ideal Taper Prediction***

# *Mold Taper Definition*



- *Mold Taper:*

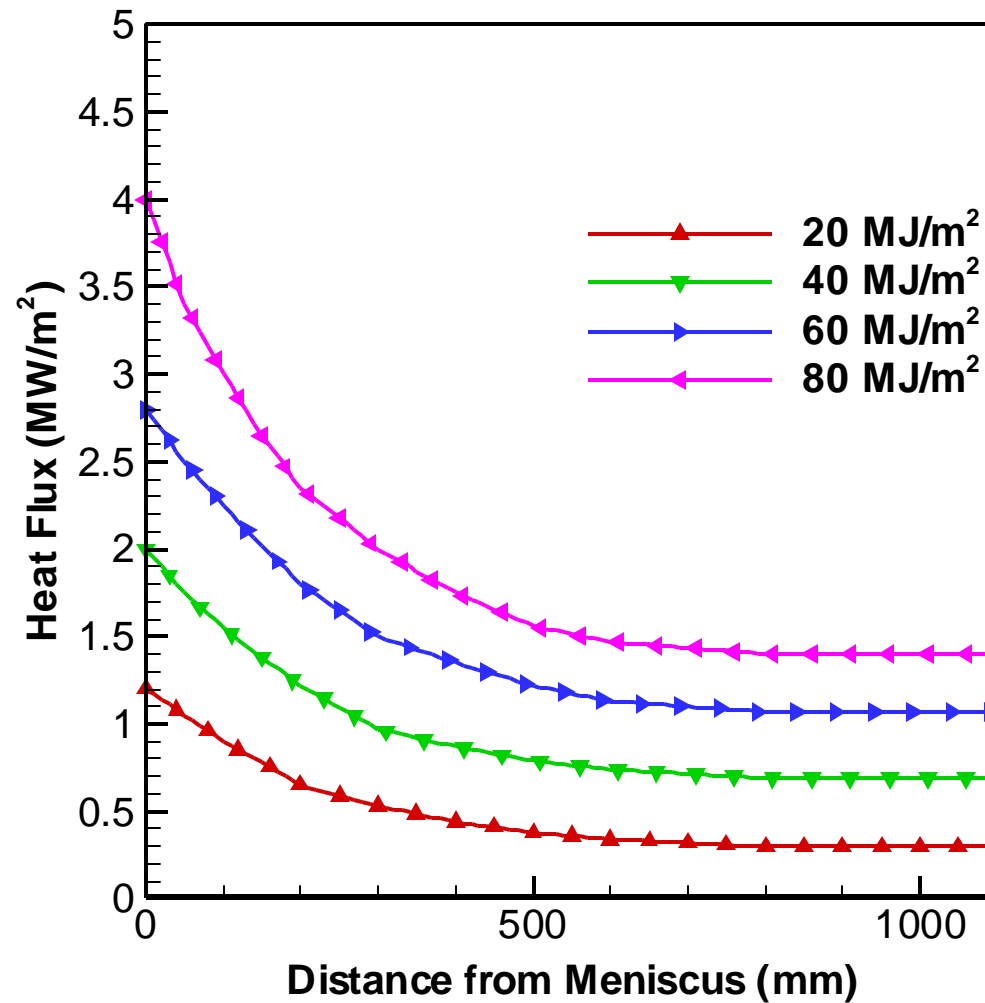
$$\%Taper = \frac{2\Delta W}{W}$$

$$\%Taper / m = \frac{2\Delta W}{LW}$$

L = Distance down mold = 1096 mm

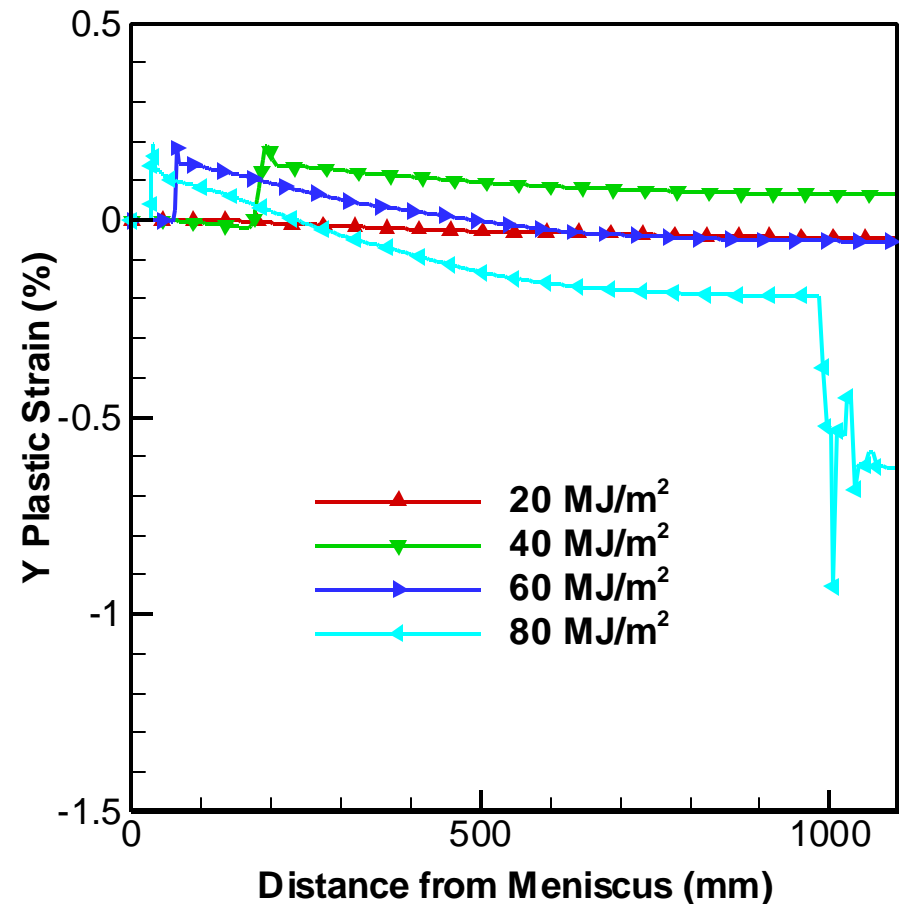
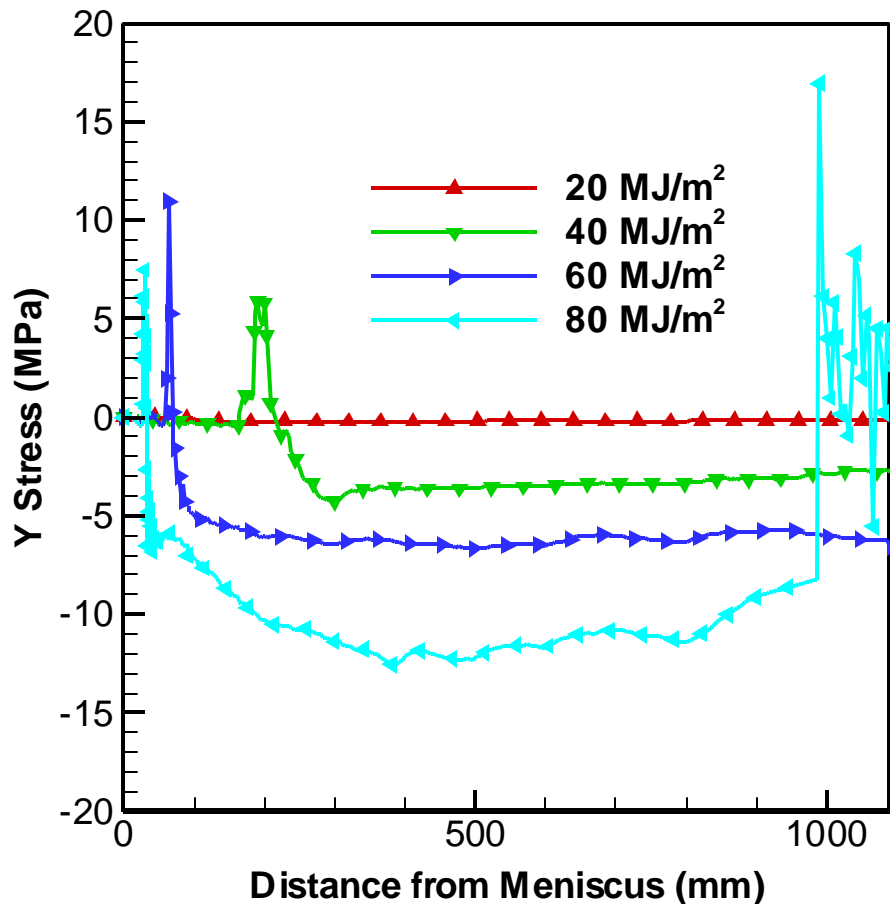
W = 960 mm

# *Heat Flux Curves used in Simulations*

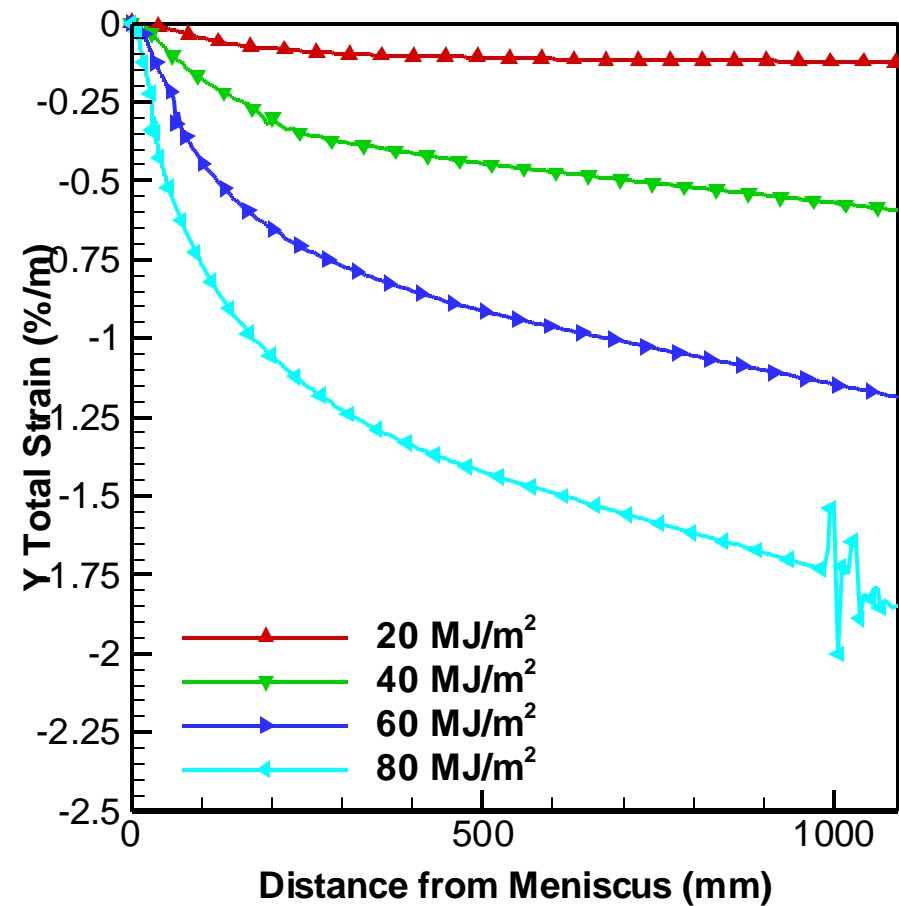
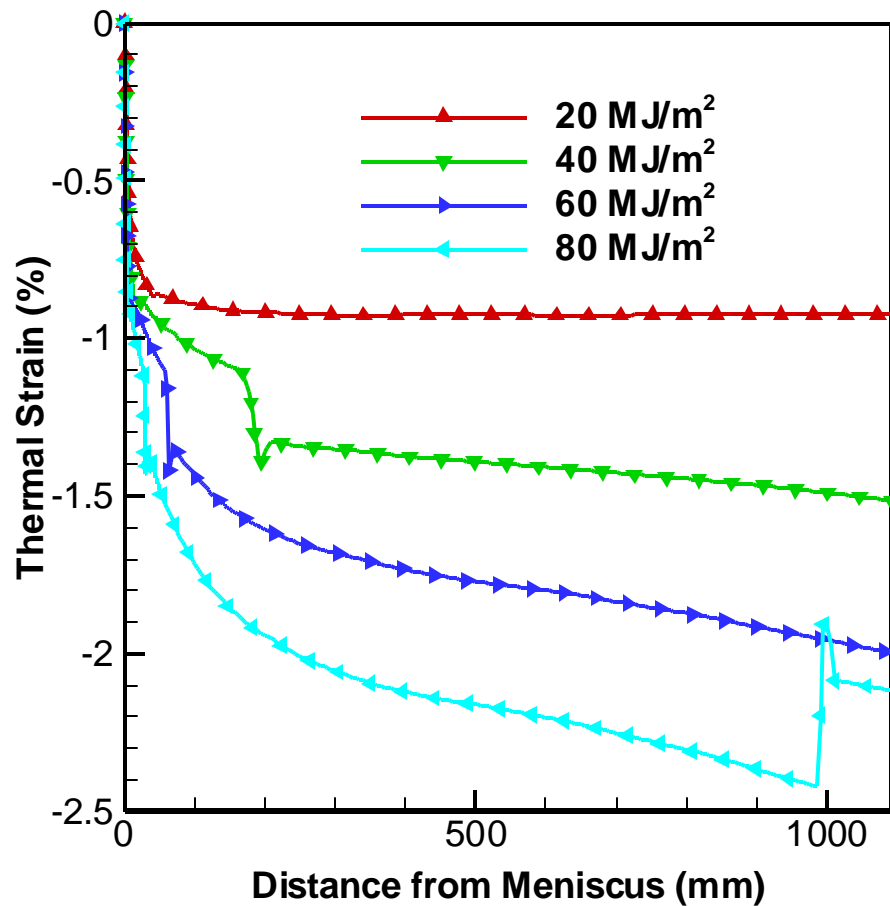




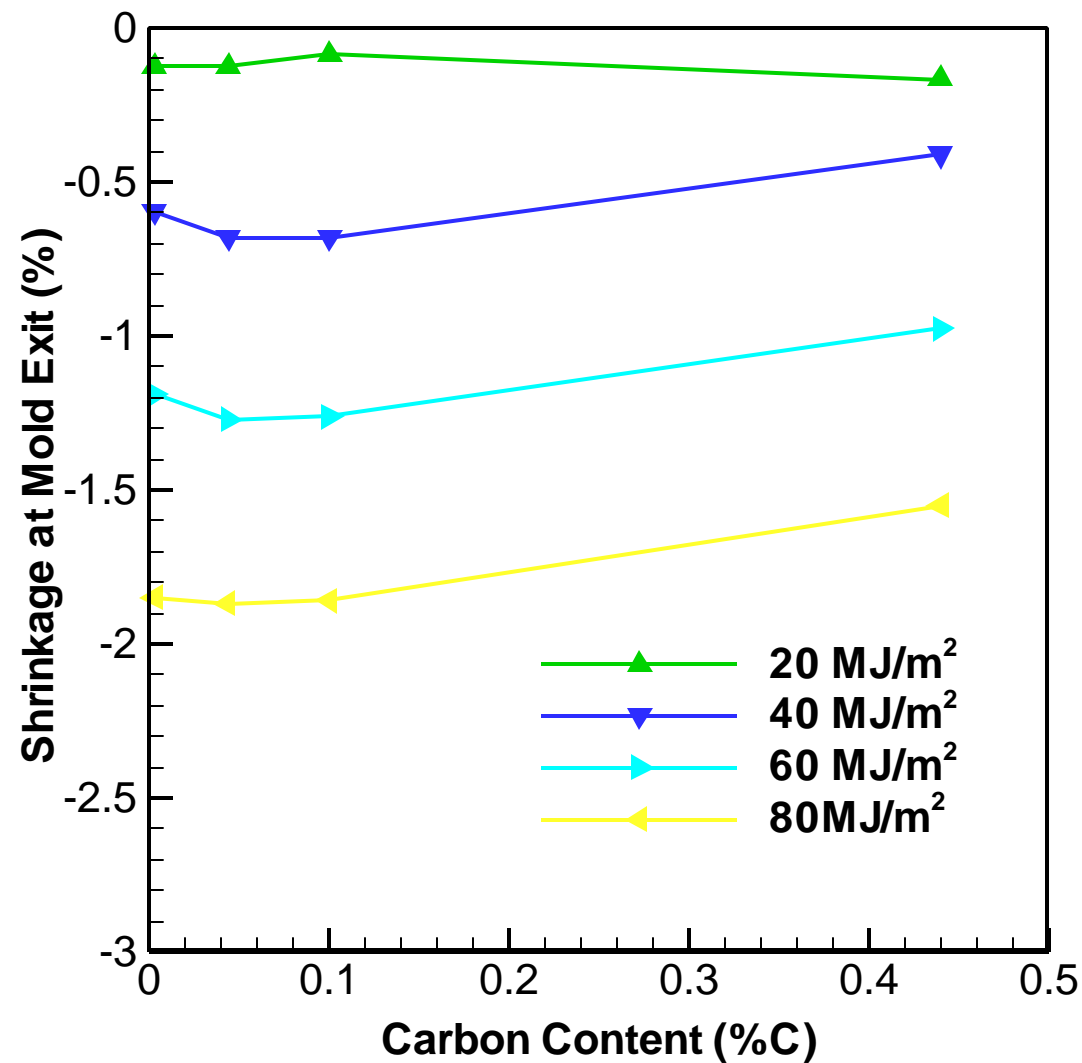
# *Stress and plastic strain histories at slab wide surface (0.003%C)*



# *Thermal strain at slab wide face and total strain (0.003%C)*

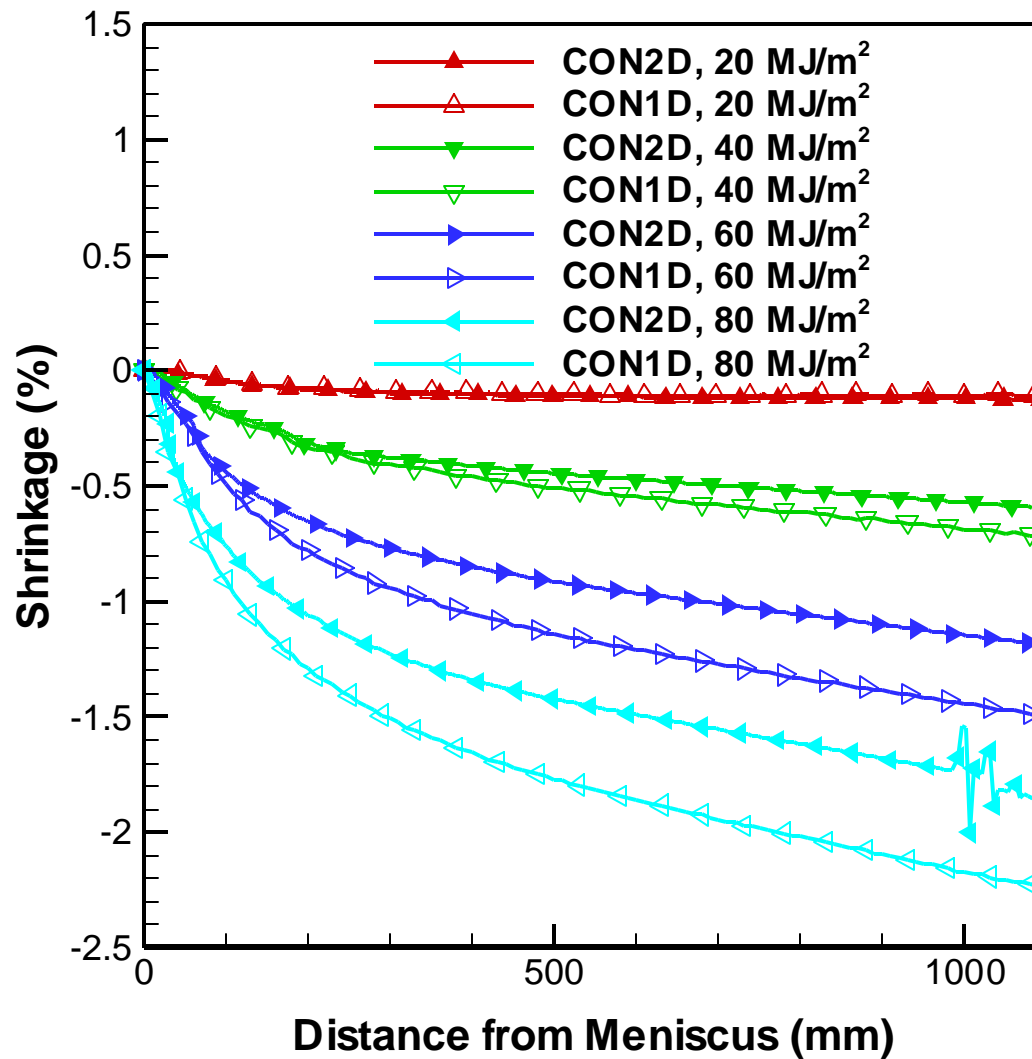


# *Carbon Content Effect*



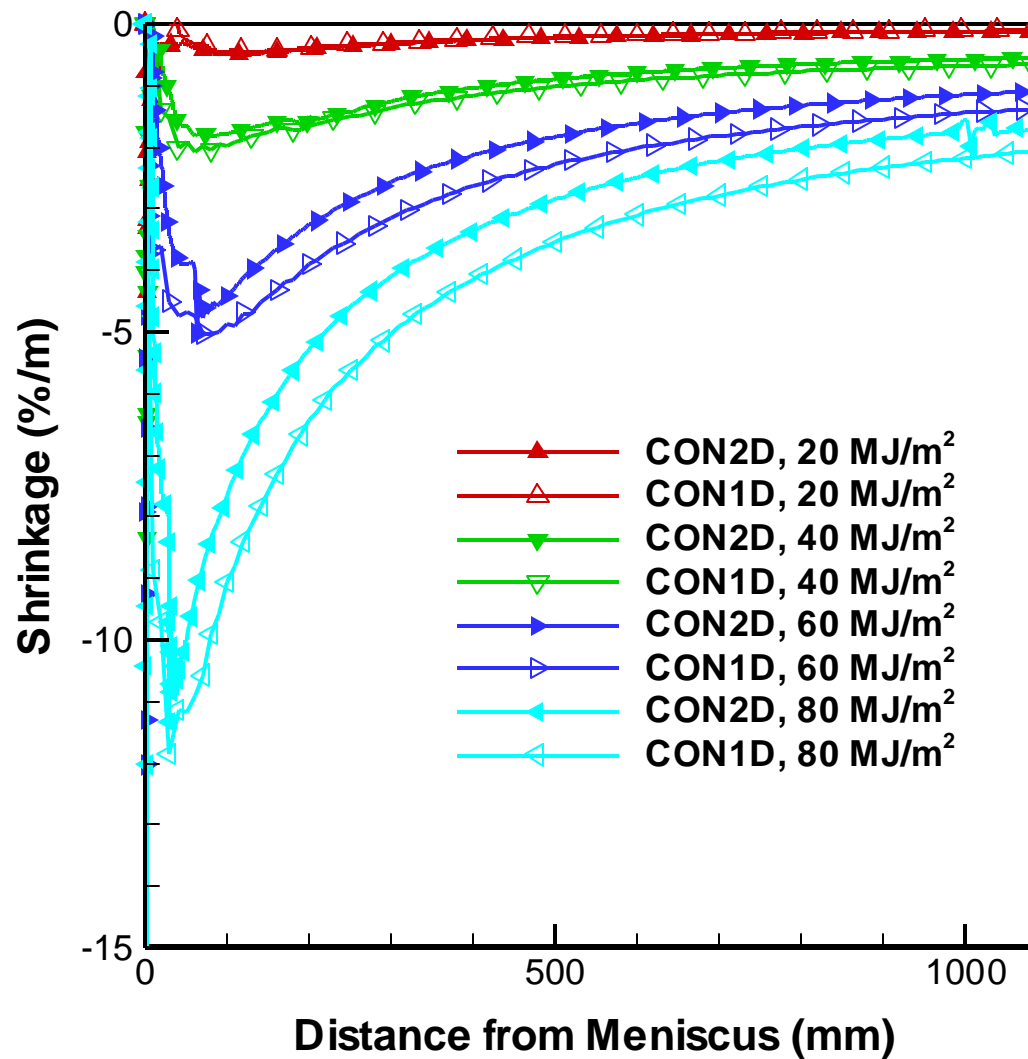
# *CON1D vs. CON2D Taper*

## *Predictions(0.003%C)*



# *CON1D vs. CON2D Predictions*

## *(cont.)(0.003%C)*

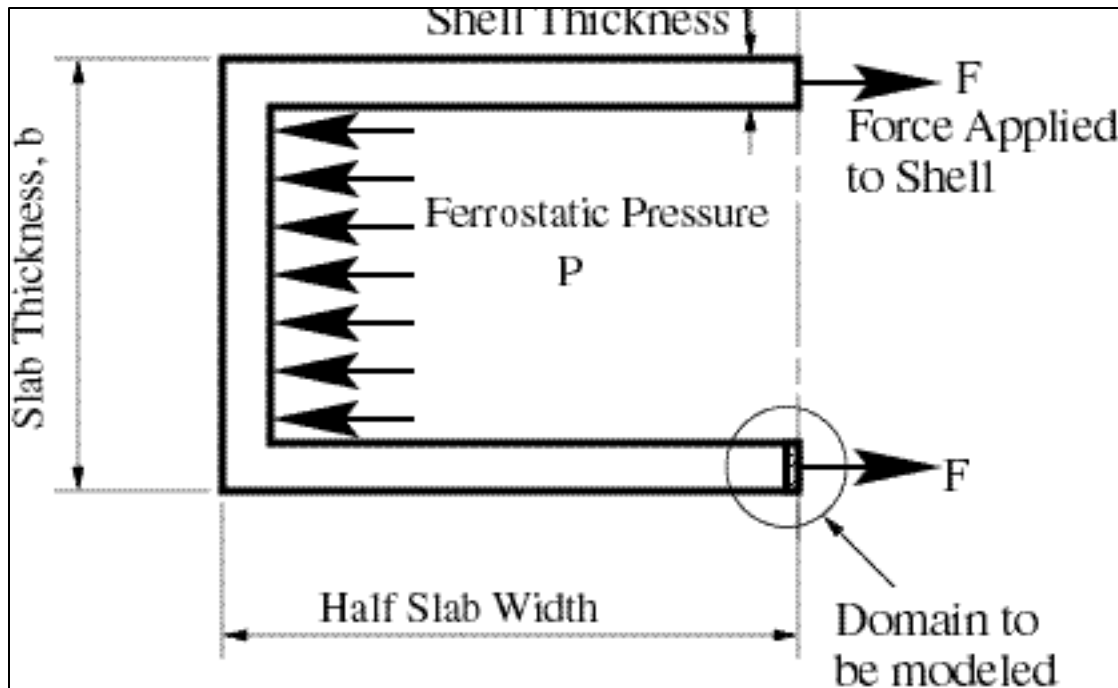


# *Conclusions*

- *Thermal strain profiles dominate the ideal taper profiles.*
- *Higher heat removal leads to larger thermal strain and larger mold taper in consequence.*
- *Phase transformation generates stress and plastic strain which have important effects on the ideal mold taper.*
- *Ideal taper is not linear. More shrinkage occurs near meniscus so ideal taper(%/m) is much larger there.*
- *The total thermal strain method prediction of ideal taper is 20% off that of the thermal stress analysis.*

# ***Critical Shell Thickness to Avoid Breakouts***

# *Force applied to the shell due to ferrostatic pressure at mold exit*



**Where:**

$$b = 132.1 \text{ mm}$$

$$H = 1096 \text{ mm},$$

$$\rho = 7800 \text{ Kg/m}^3$$

$$g = 9.81 \text{ m/s}^2,$$

$$t = 2 \sim 17 \text{ mm}$$

Force balance in horizontal direction yields:

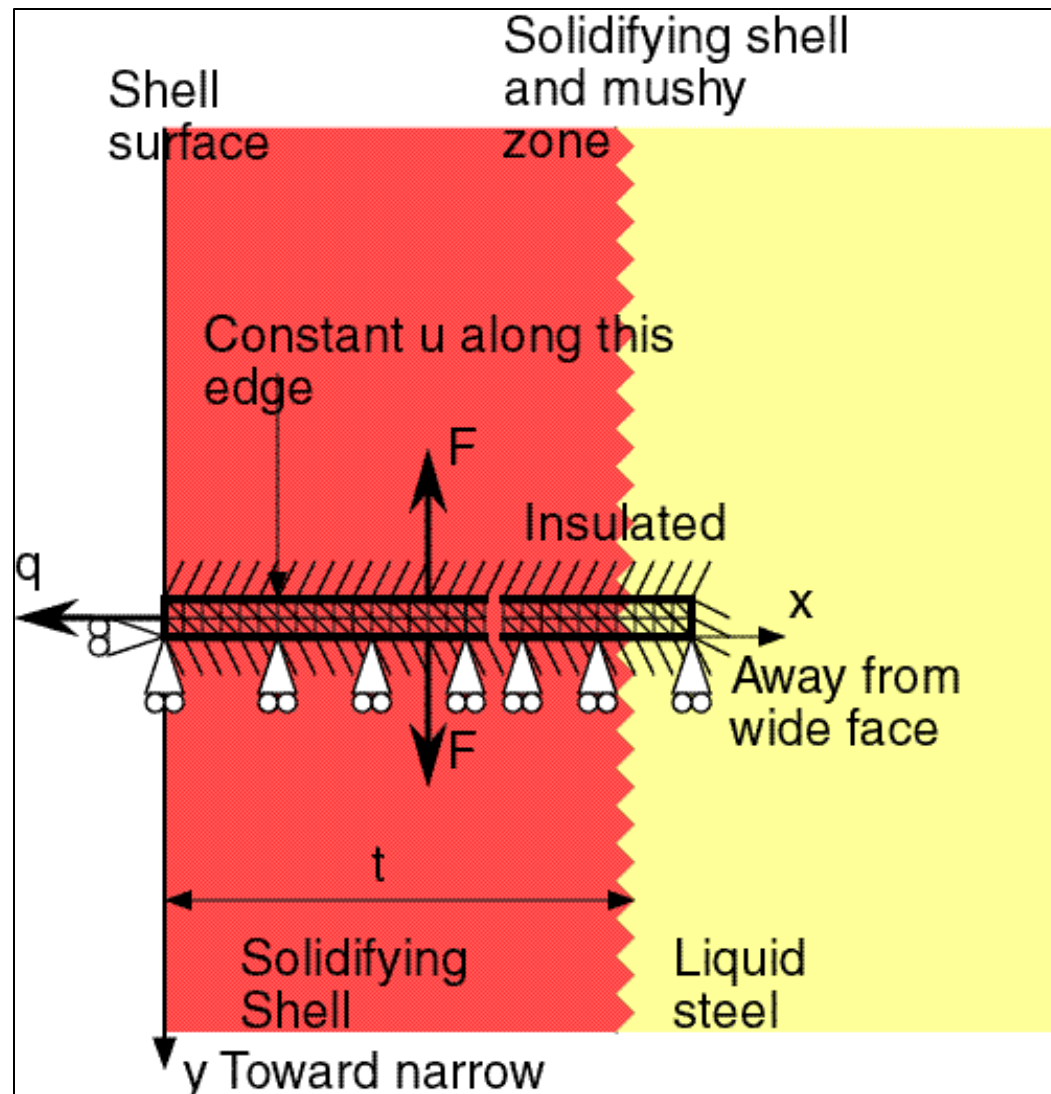
$$\mathbf{F = \rho g H(b-2t)/2 = 4 \text{ (KN/m)}^*}$$

- Values of the parameters are based on ARMCO caster in Mansfield, OH

\* Constant force value is chosen to make this parametric study convenient



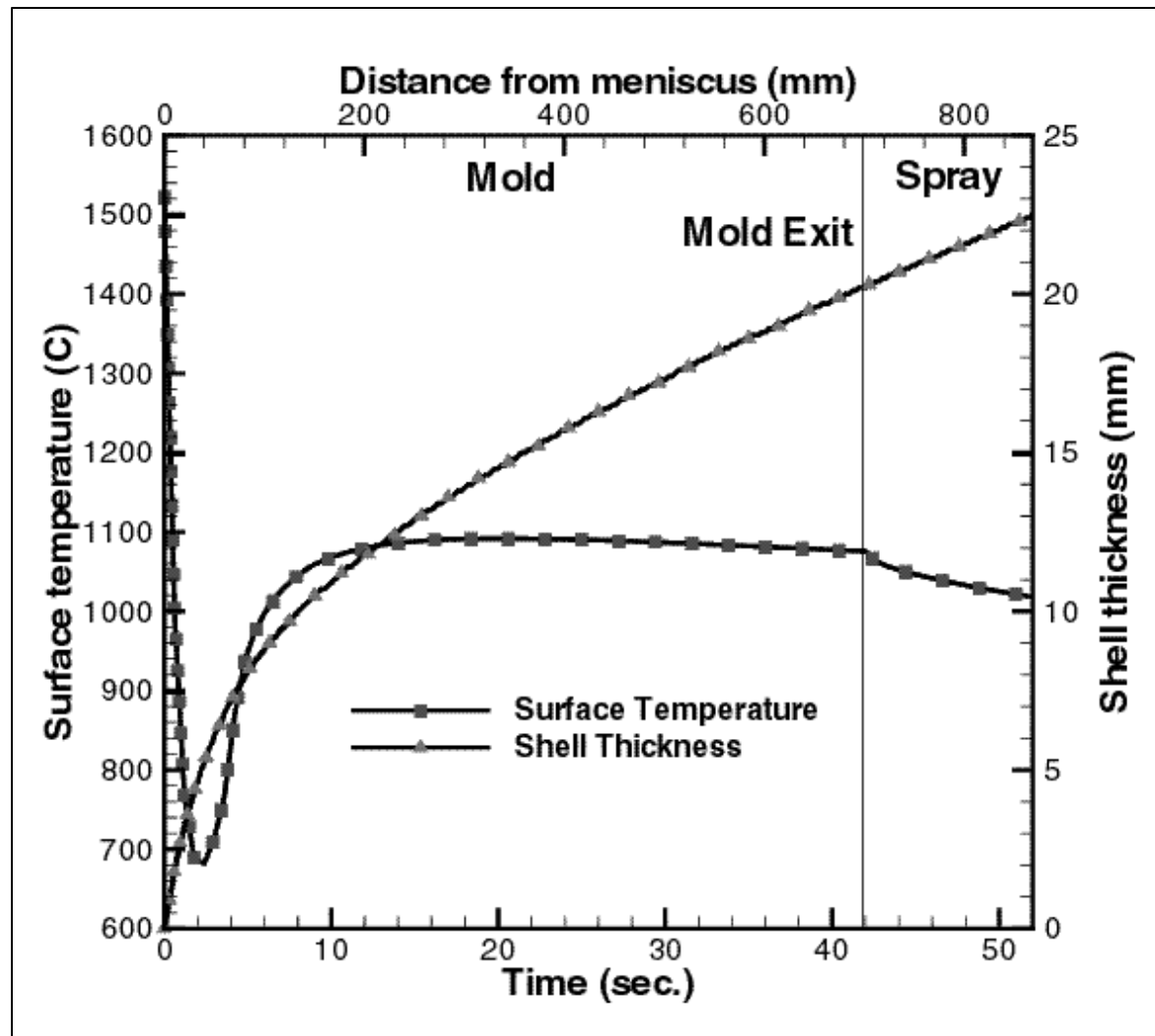
# Modeling Domain



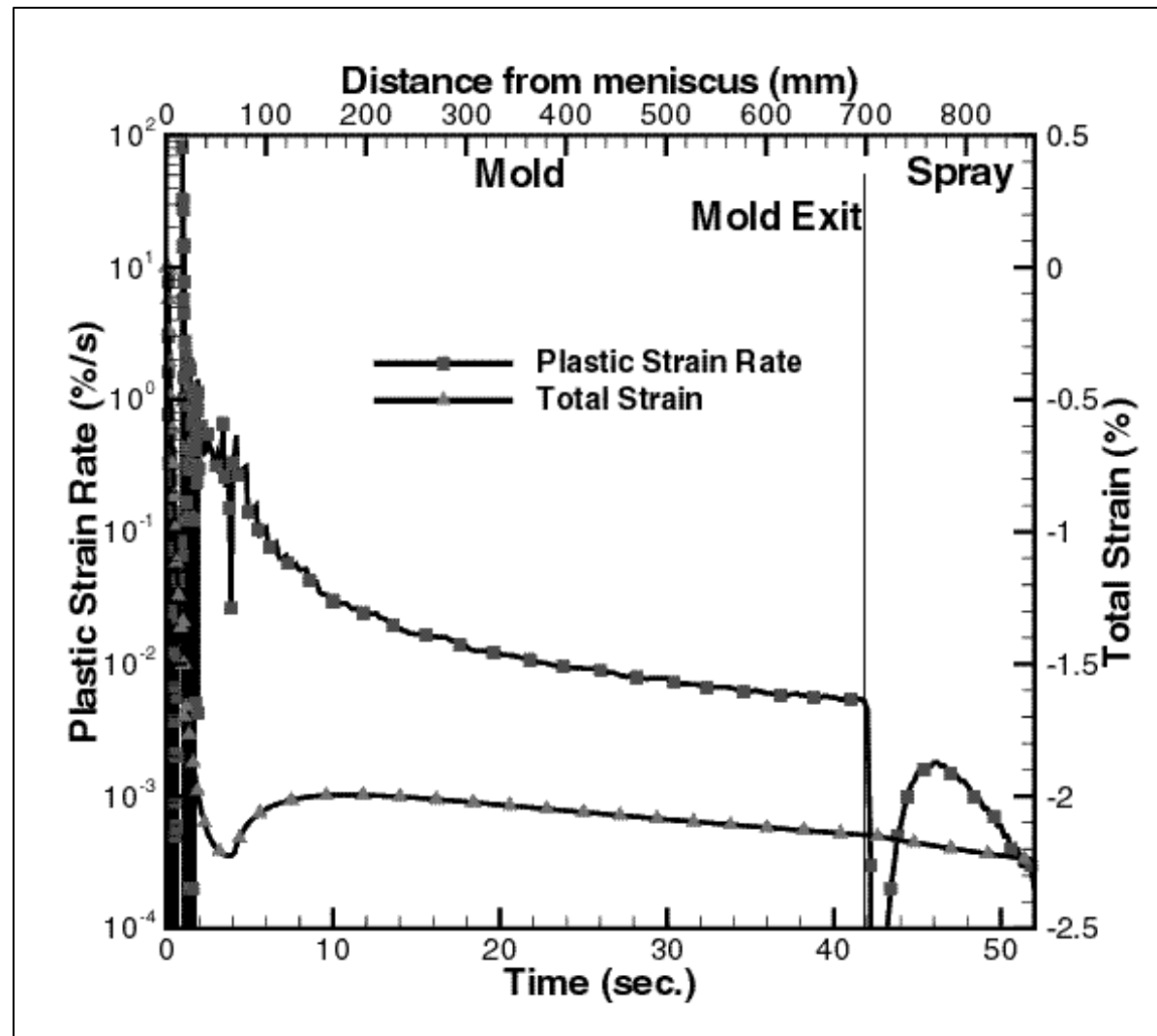
# *Sample casting conditions*

- *Casting speed : 1.0 ~ 160 m/min (16.67 ~ 2666.67mm/sec.)*
- *Pouring temp.:  $T_{liquidus} + \text{Superheat}(1\text{ }^{\circ}\text{C or } 50\text{ }^{\circ}\text{C})$*
- *Bloom section size : 50, 100, 200, 300, 400 mm*
- *Working mold length: 300, 500, 700, 900, 1100 mm*
- *Mold flux -- solidification temp.: 1193.0  $^{\circ}\text{C}$* 
  - viscosity at 1300  $^{\circ}\text{C}$ : 0.7 poise*
  - local consumption rate: 0.255  $\text{Kg/m}^2$*

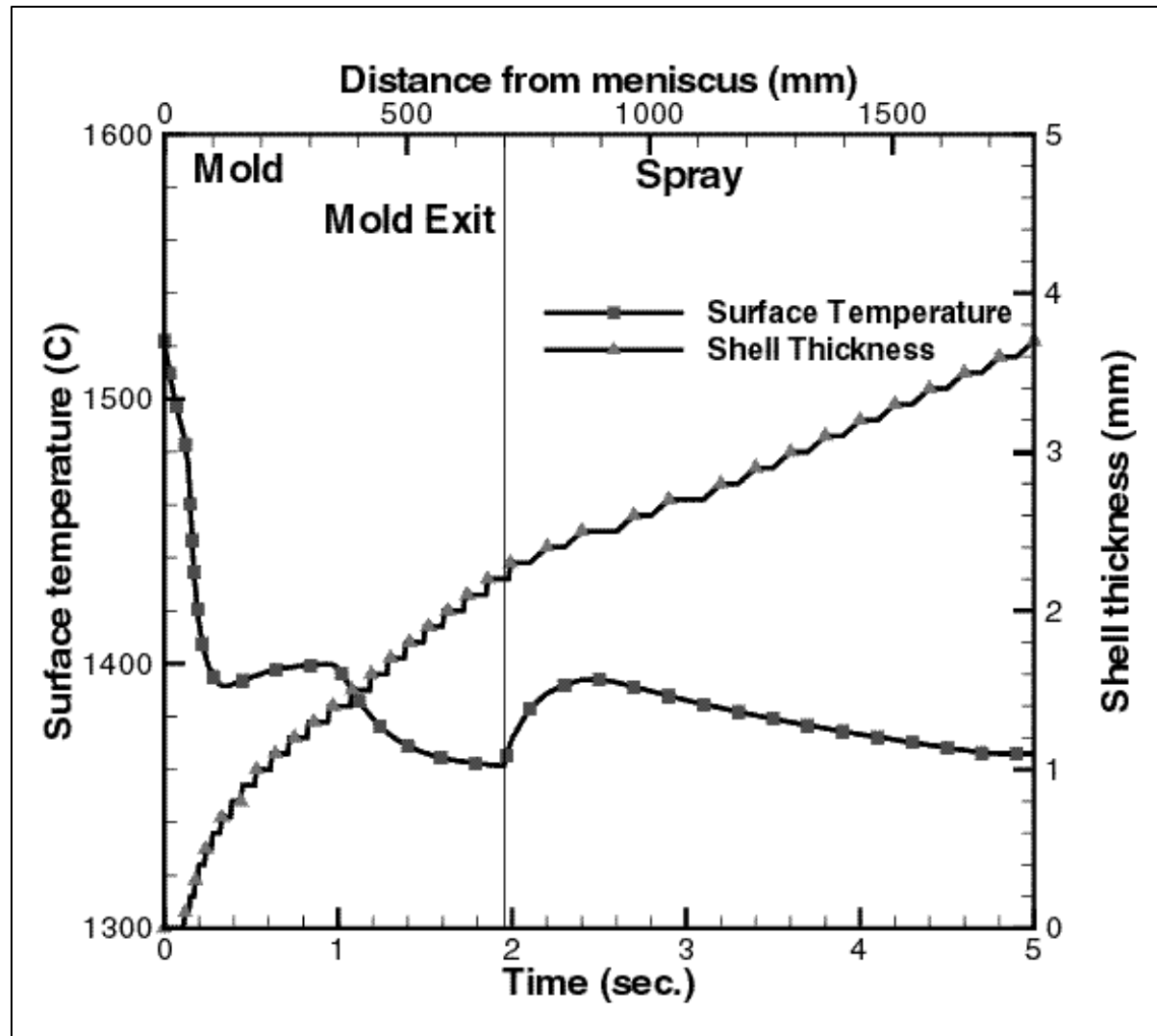
# *Surface Temperature and shell thickness histories (casting speed: 1 m/min)*



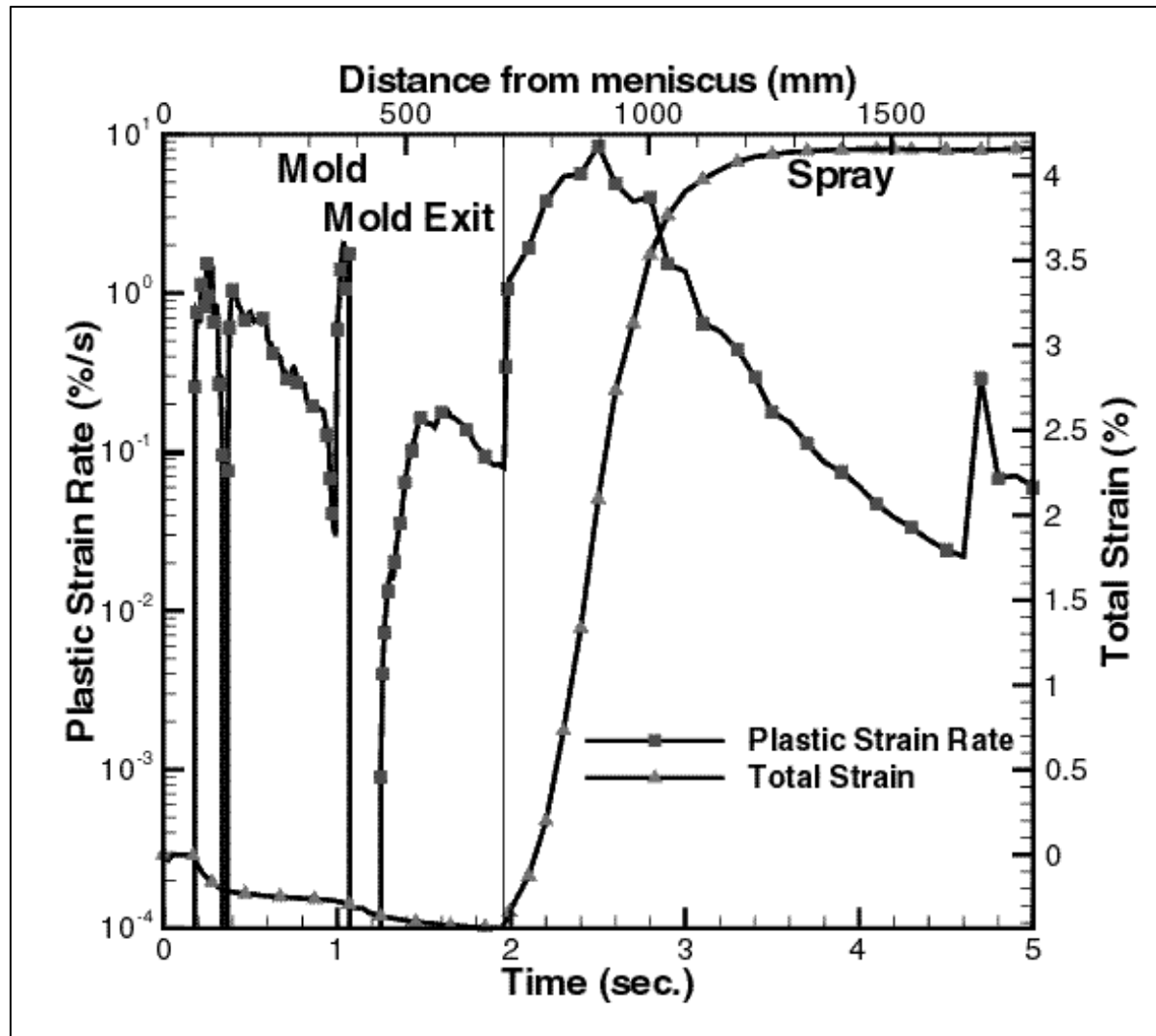
# *Total strain and plastic strain rate histories (casting speed: 1m/min)*



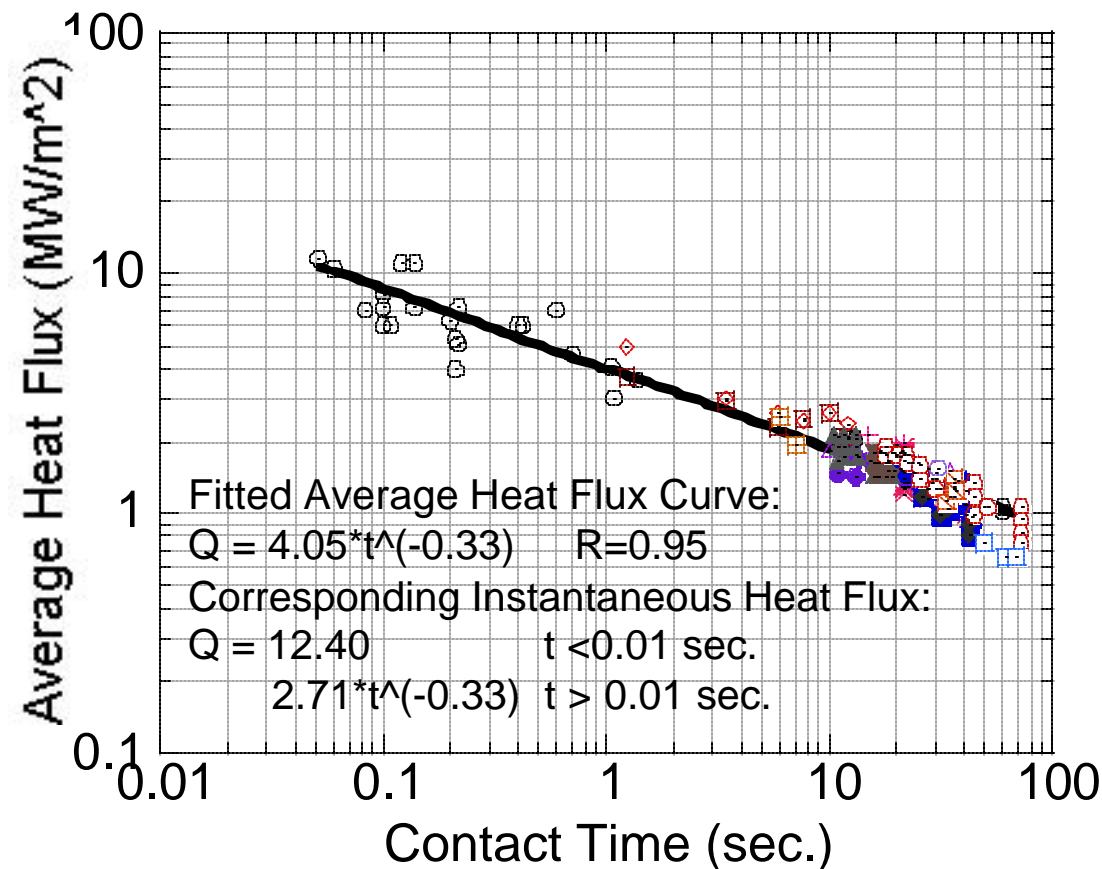
# *Surface temperature and shell thickness histories (casting speed: 21.5 m/min)*



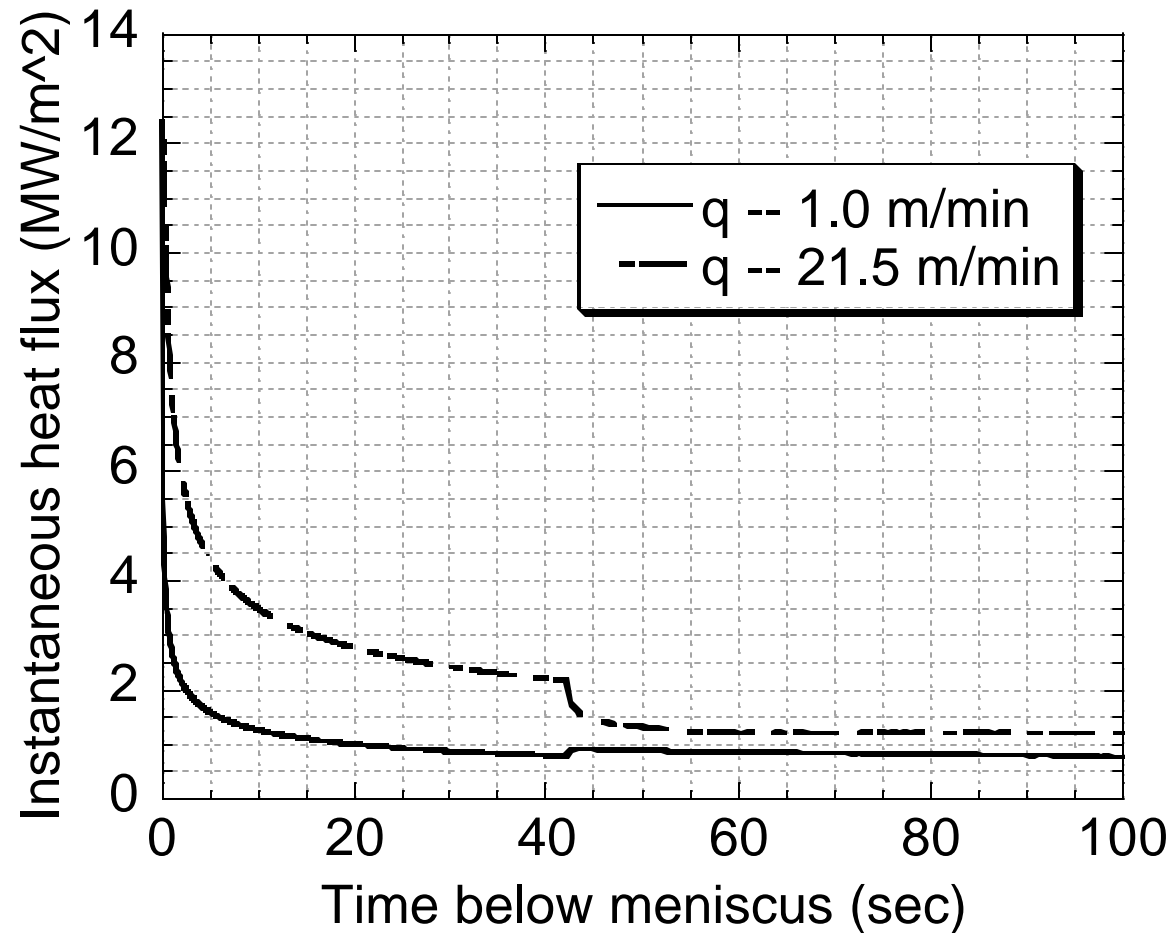
# *Total strain and plastic strain rate histories (casting speed: 21.5 m/min)*



# Average Heat Flux vs Contact Time

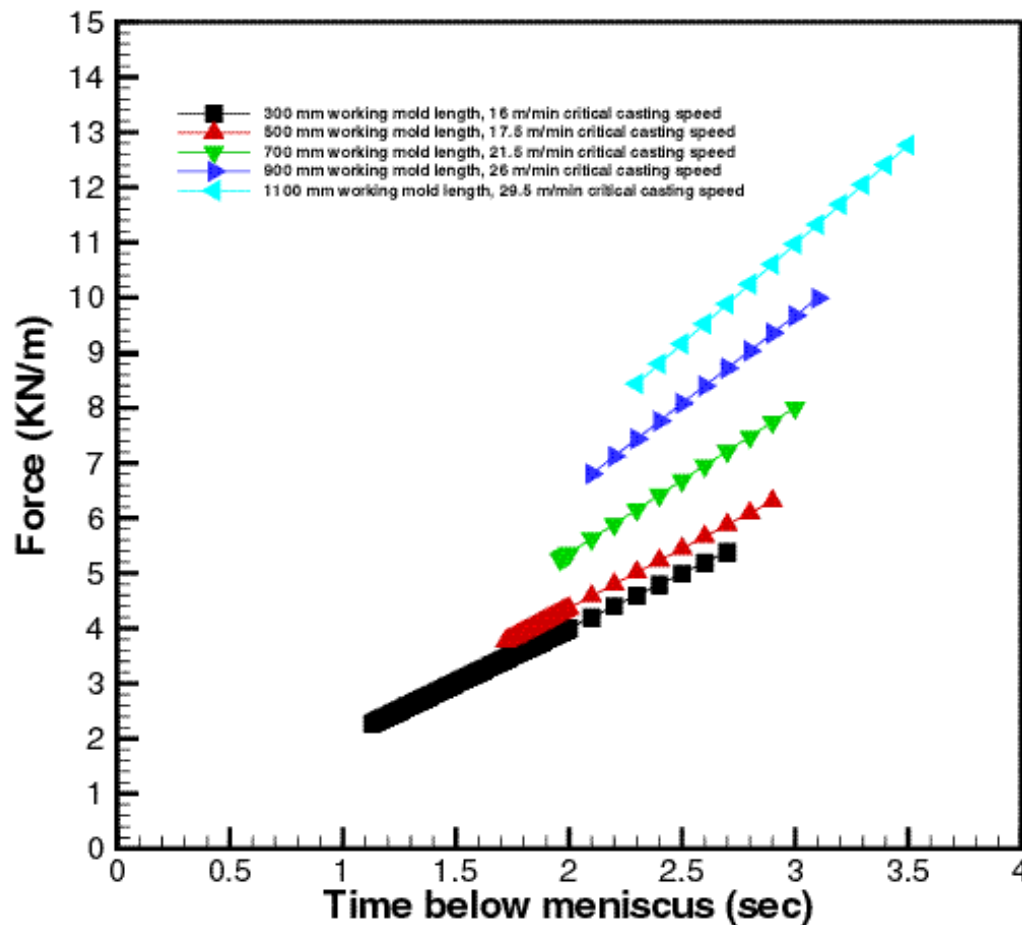


# *Instantaneous Heat Flux*





# *Critical Force Leading to Break Shell vs Applied Time*



1. 0.044%C Plane carbon steel
2. Line segments begin at the time beginning to apply the force and end at break point

# *Critical shell thickness based on fracture criterions from Young Mok WON et. al.*

%C	Shell thickness at 1% strain (mm)	Fracture Strain (%) <sup>*</sup>	Critical Shell Thickness (mm)			Strain Rate (1/s)		Heat Removed At Mold Exit (MJ/m <sup>2</sup> )	Surf. Temp. At Mold Exit (Degree C)
			Total (mm)	Solid Layer	Mushy Layer (mm)	At mold exit	>10 sec		
<b>0.003</b>	<b>5.8</b>	<b>6.1</b>	<b>3.5</b>	<b>3.1mm d</b>	<b>0.4</b>	<b>8.80E-03</b>	<b>4.86E-04</b>	<b>14.89</b>	<b>1466.86</b>
<b>0.044</b>	<b>6.1</b>	<b>4.5</b>	<b>4.2</b>	<b>3.8mm g+d</b>	<b>0.7</b>	<b>9.70E-03</b>	<b>6.10E-04</b>	<b>17.07</b>	<b>1444.35</b>
<b>0.1</b>	<b>4.6</b>	<b>3.4</b>	<b>4.1</b>	<b>3.6mm g+d</b>	<b>0.9</b>	<b>7.70E-03</b>	<b>1.35E-05</b>	<b>19.35</b>	<b>1421.43</b>
<b>0.44</b>	<b>2.9</b>	<b>1.8</b>	<b>2.8</b>	<b>0.7mm g</b>	<b>2.1</b>	<b>5.20E-03</b>	<b>2.76E-07</b>	<b>23.11</b>	<b>1360.00</b>
<b>304 SS</b>	<b>1.5</b>	<b>1.3</b>	<b>1.6</b>	<b>0.7mm g</b>	<b>0.9</b>	<b>5.19E-02</b>	<b>5.00E-06</b>	<b>13.40</b>	<b>1397.05</b>
<b>430 SS</b>	<b>6.1</b>	<b>3.1</b>	<b>3.4</b>	<b>3.1mm d</b>	<b>0.3</b>	<b>5.50E-02</b>	<b>1.00E-06</b>	<b>9.69</b>	<b>1470.28</b>

*\* Critical fracture strain is calculated based on the empirical equation from the new by WON et. al. which is to be published in Met. Trans.*

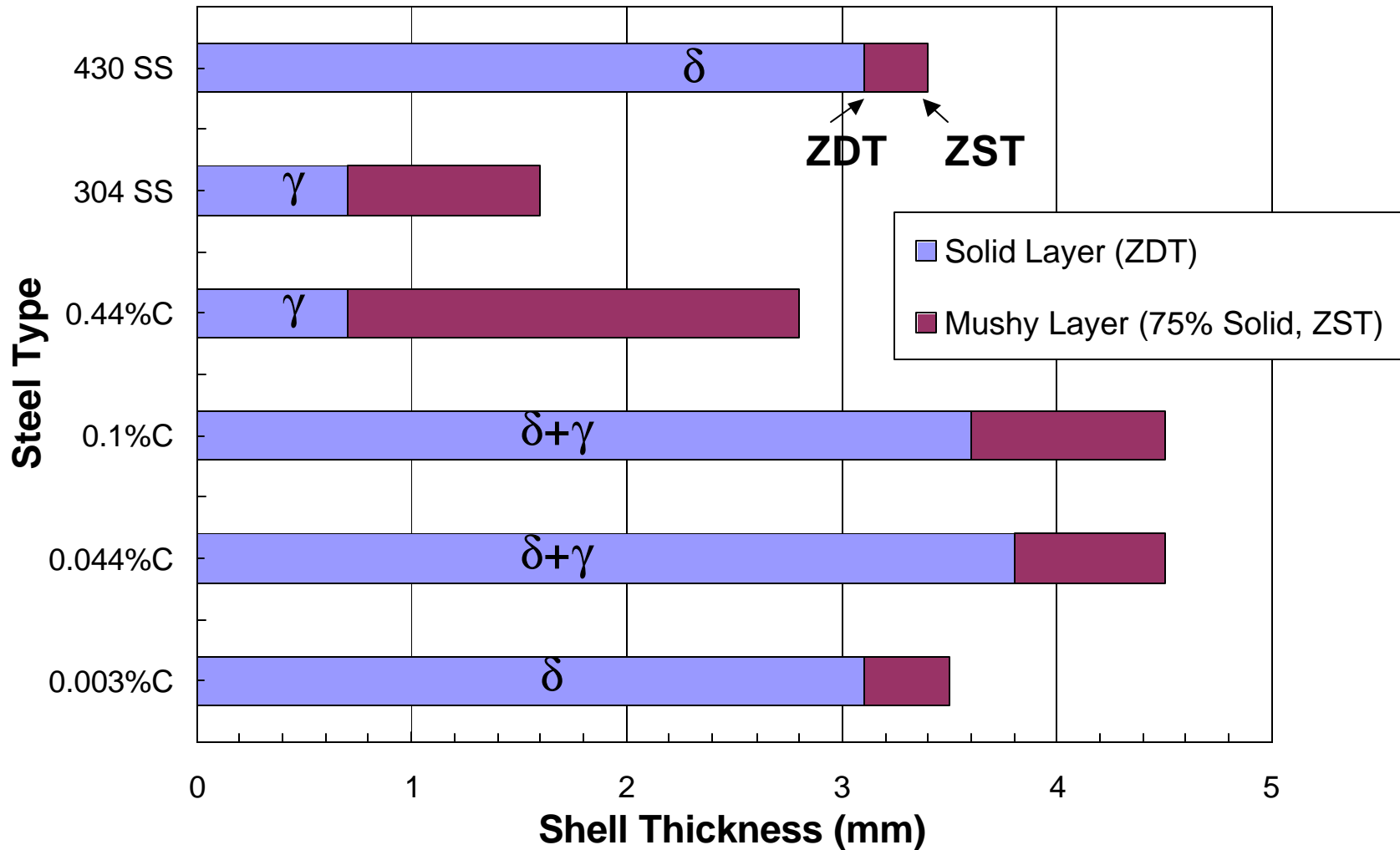
$$e_c = \frac{j}{\dot{\epsilon}^{m^*} \Delta T_B^{n^*}}$$

where :

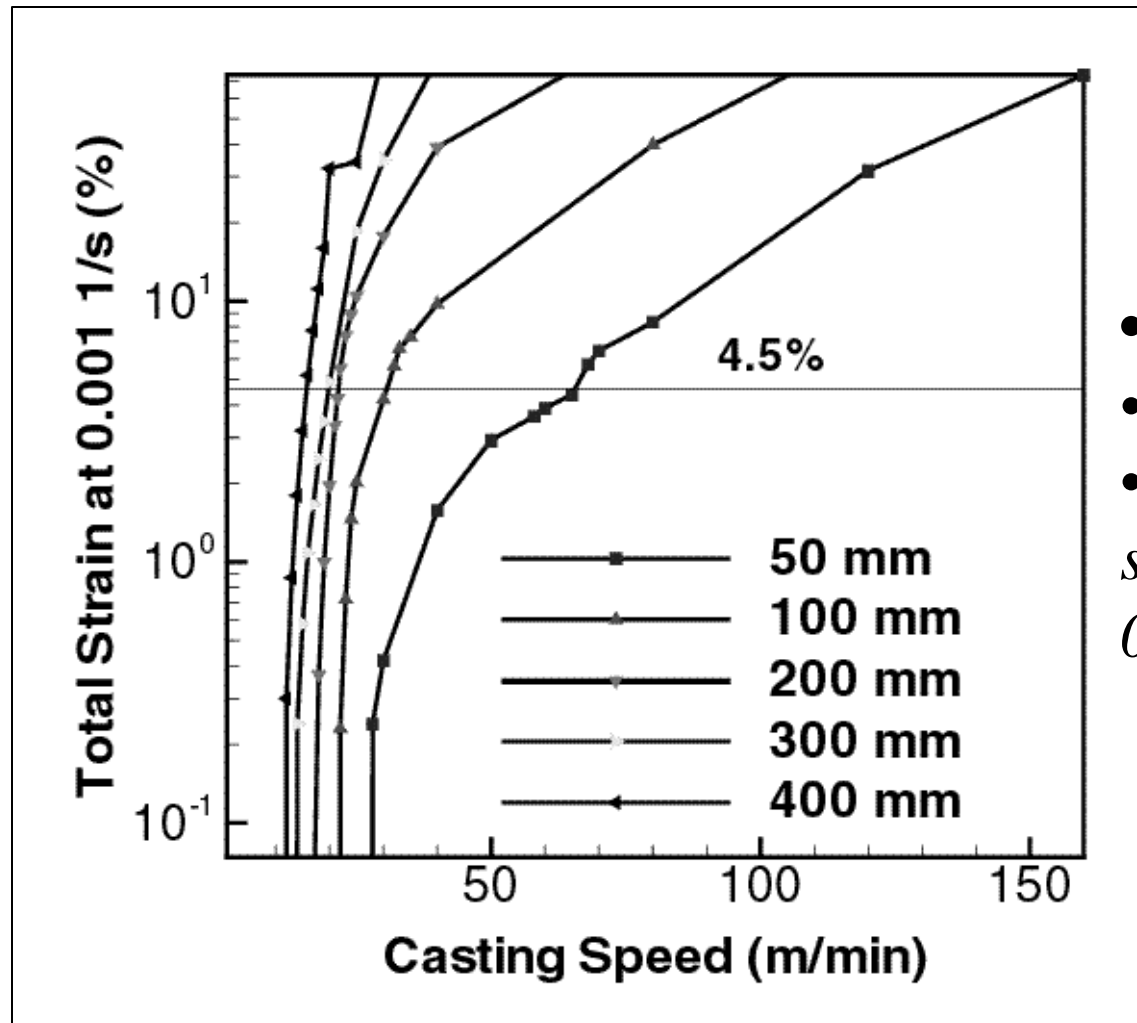
$$\Delta T_B = T(f_s = 0.9) - T(f_s = 0.99)$$

$$m^* = 0.3131, n^* = 0.8638, j = 0.02821$$

# Critical Shell Thickness Structures



# *Critical casting speed*



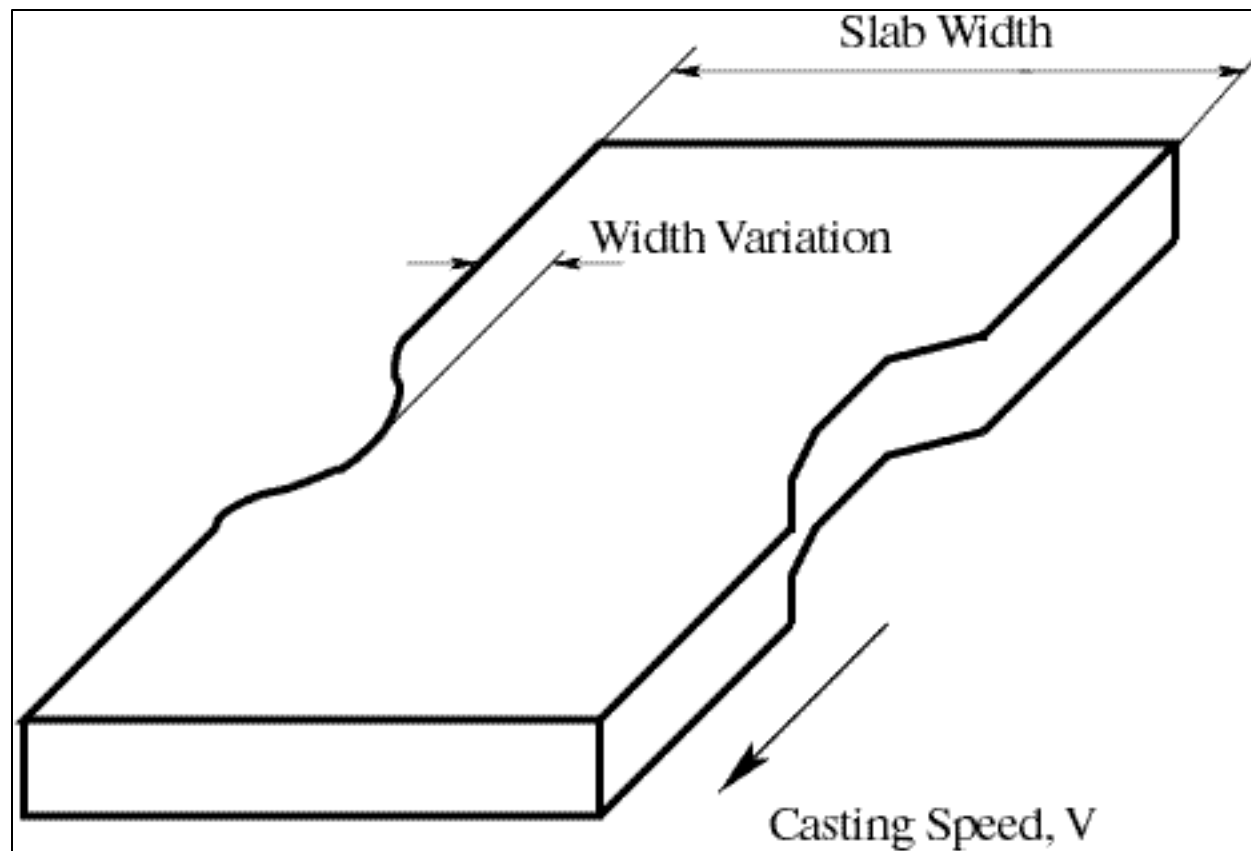
- *0.044%C carbon steel*
- *1 °C superheat*
- *measured when plastic strain rate drops to  $0.001 \text{ s}^{-1}$*

# *Conclusions*

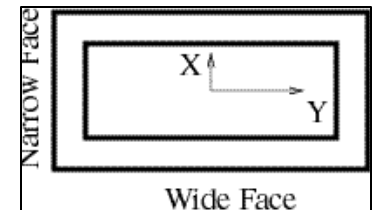
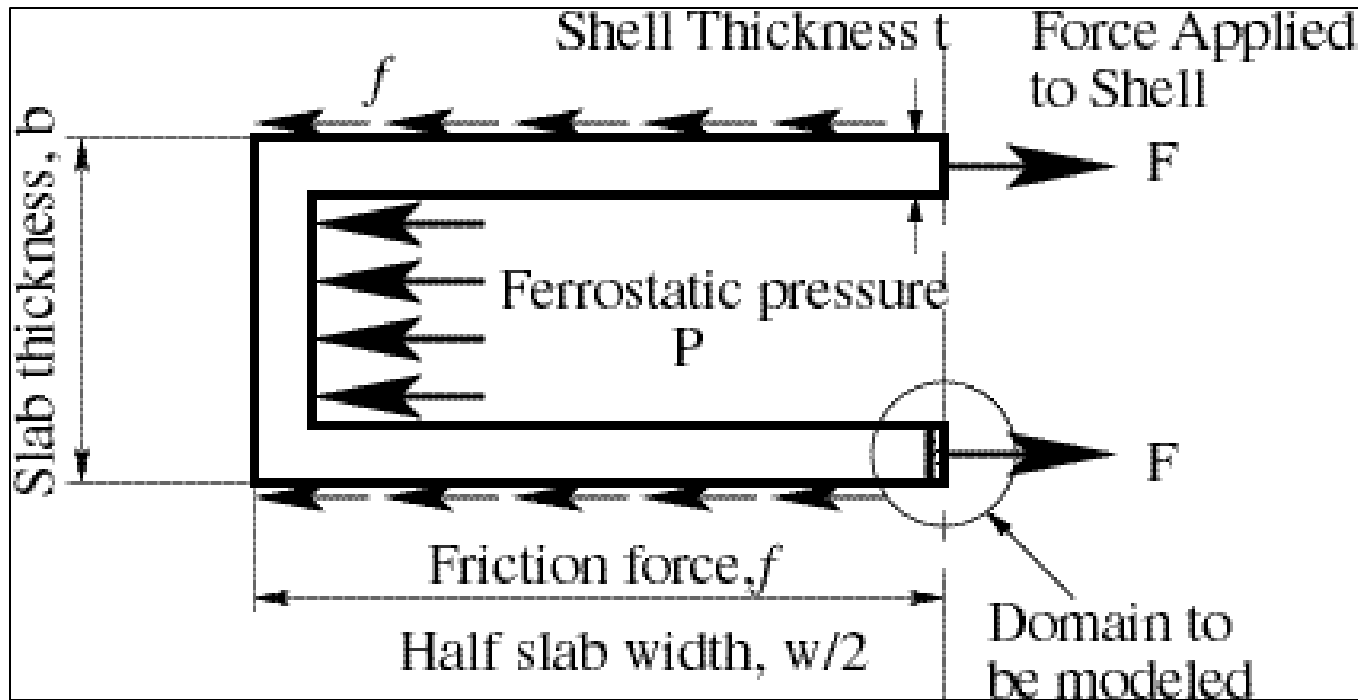
- *d phase is weaker than g phase so lower carbon steel is weaker and requires a larger shell thickness at mold exit for 1% strain.*
- *Steel is more brittle with increasing carbon content.*
- *Combining the last two statements, it can be predicted that most breakout susceptible steel is the low carbon steel(0.04%C ~ 0.15%C).*
- *Considering uneven shell growth in 0.1%C steel, it is most likely to have thin spots leading to breakouts.*
- *Superheat does not affect the critical shell thickness.*

# *Prediction of Strand Width Variations*

# *Width Variation*



# *A potential mechanism caused width variation*



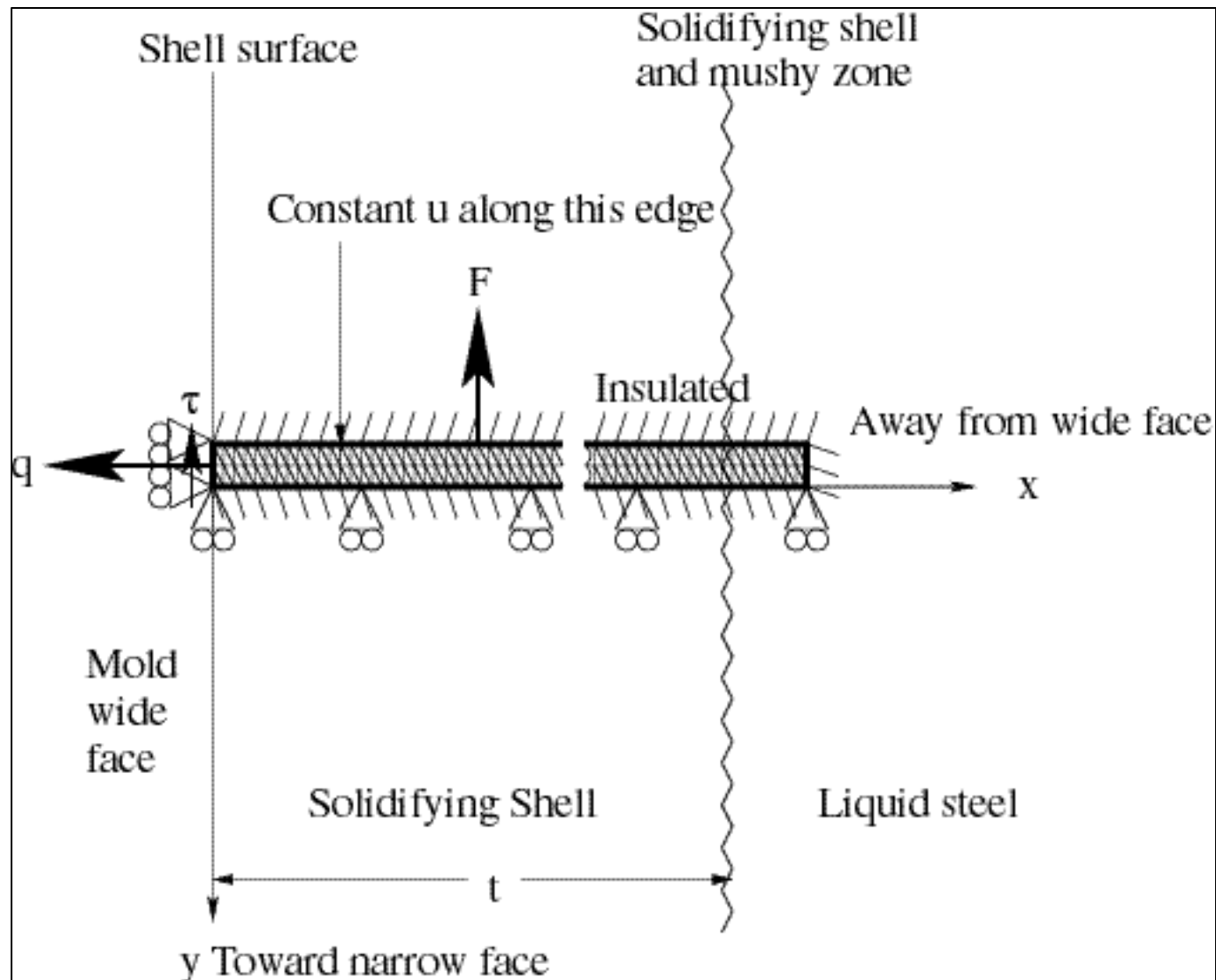
$$f(N) = \mathbf{f} r g h \left( \frac{w}{2} \right) dz$$

$$F(N) = \frac{P(b-2t)dz}{2} + f$$

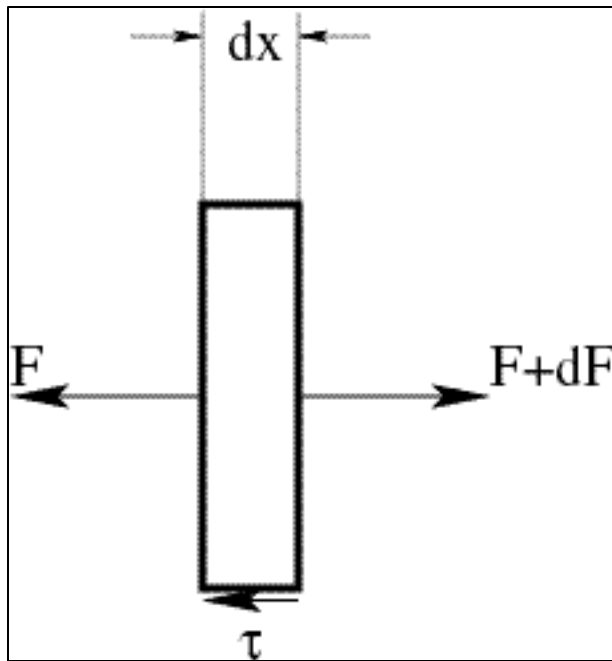
where  $\mathbf{f}$  is coefficient of friction



## *Modeling domain*



# *Force applied in this study*



$$t \text{ (MPa)} = \left( \frac{f}{\frac{w}{2} dz} \right) = \mathbf{f} r g h$$

$$F \text{ (N / m)} = \frac{r g h (b - 2t)}{2} + \mathbf{f} r g h x$$

- *X is the distance between wide face slice being modeled and narrow face.*
- *Force is per unit length in z-direction*

$$dF = t dx \Rightarrow \int_{F_0}^F dF = \int_0^x t dx \Rightarrow F = F_0 + tx$$

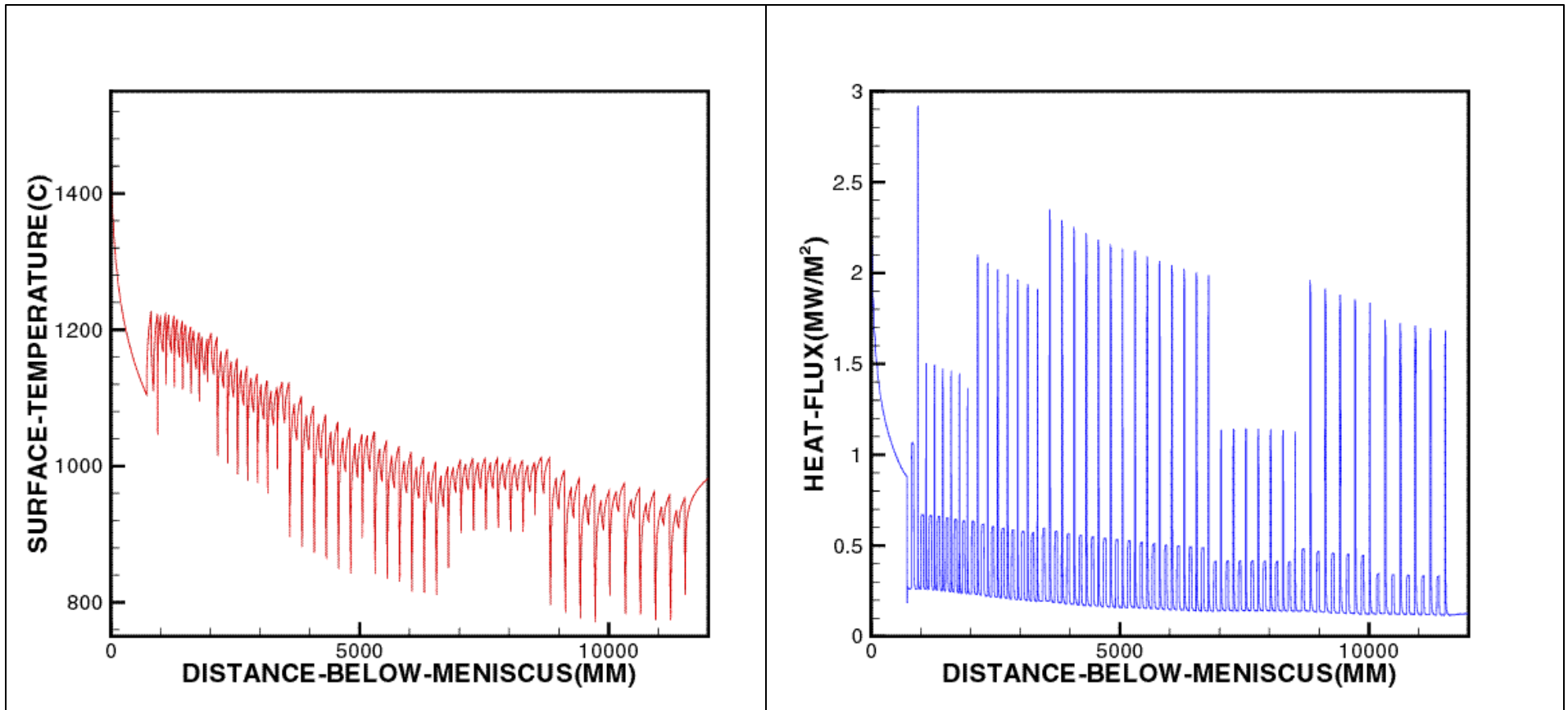
# *Parameters used in this study\* \*\**

<i>Mold Width (mm)</i>	1120
<i>Mold Thickness (mm)</i>	203.3
<i>Working mold length (mm)</i>	729.4118
<i>Friction coefficient between slab and rollers</i>	0.3
<i>Casting Speed (m/min.)</i>	0.9144
<i>Tundish Temperature ( C )</i>	1550
<i>Liquidus Temperature ( C )</i>	1502
<i>Solidus Temperature ( C )</i>	1477
<i>Density (Kg/m<sup>3</sup>)</i>	7800
<i>Gravity acceleration (m/sec<sup>2</sup>)</i>	9.8
<i>Modeling domain width (mm)</i>	100
<i>Modeling domain thickness (mm)</i>	0.2
<i>Element number along width[heat/stress]</i>	1000/500
<i>Element number along thickness[heat/stress]</i>	2/1

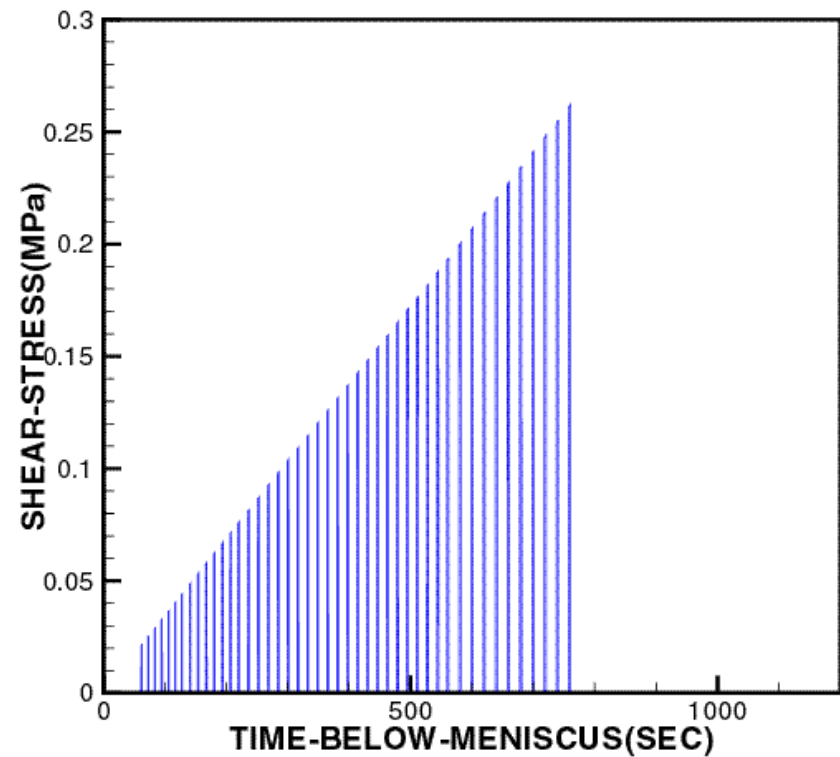
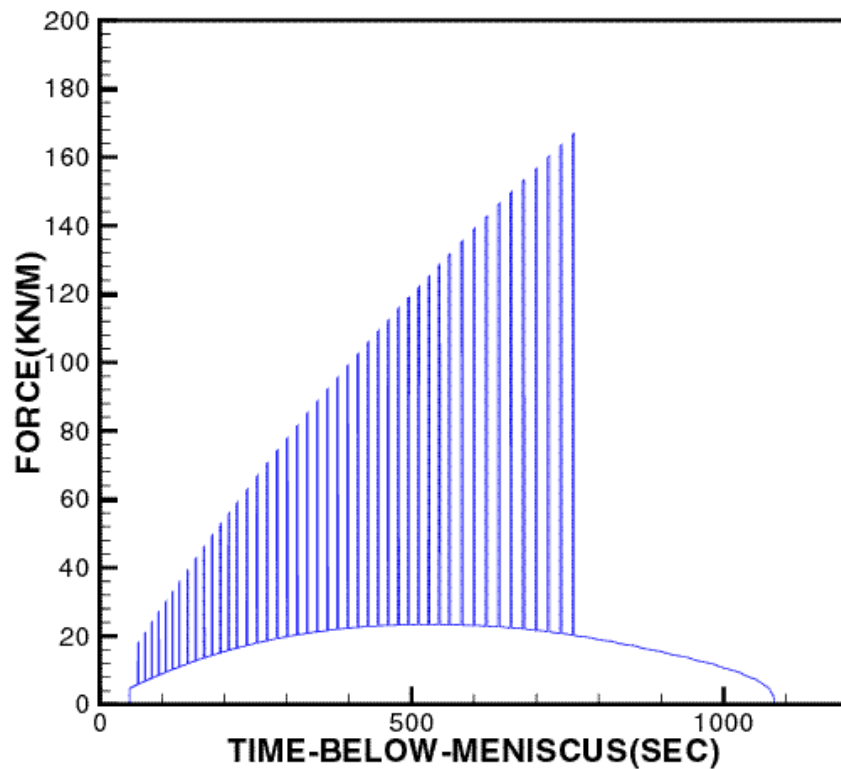
\* Conditions are measurements at ARMCO from Jay Watson, 1995

\*\* Steel is 409 stainless steel (d phase only)

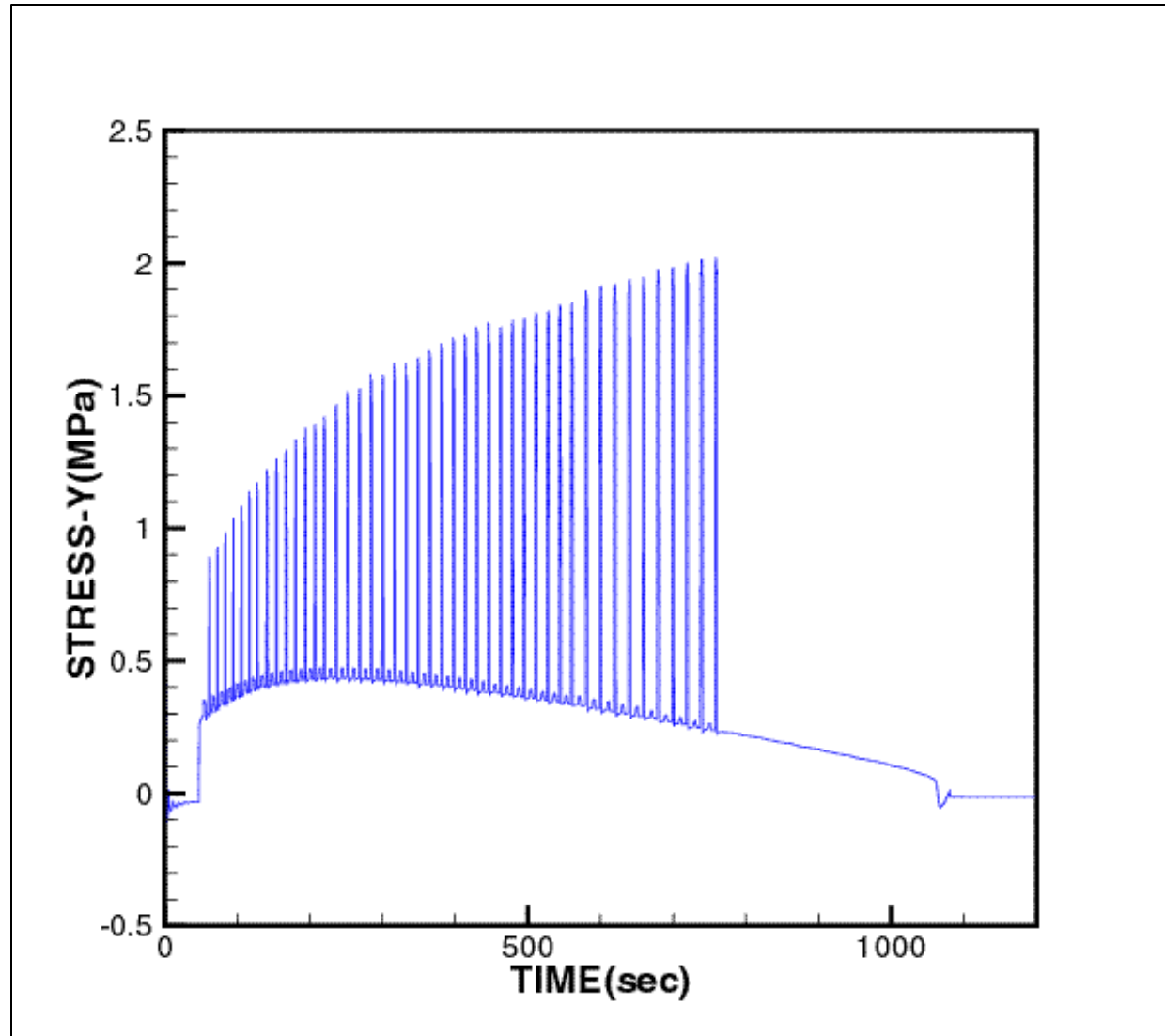
# *Surface temperature and heat flux from CON1D*



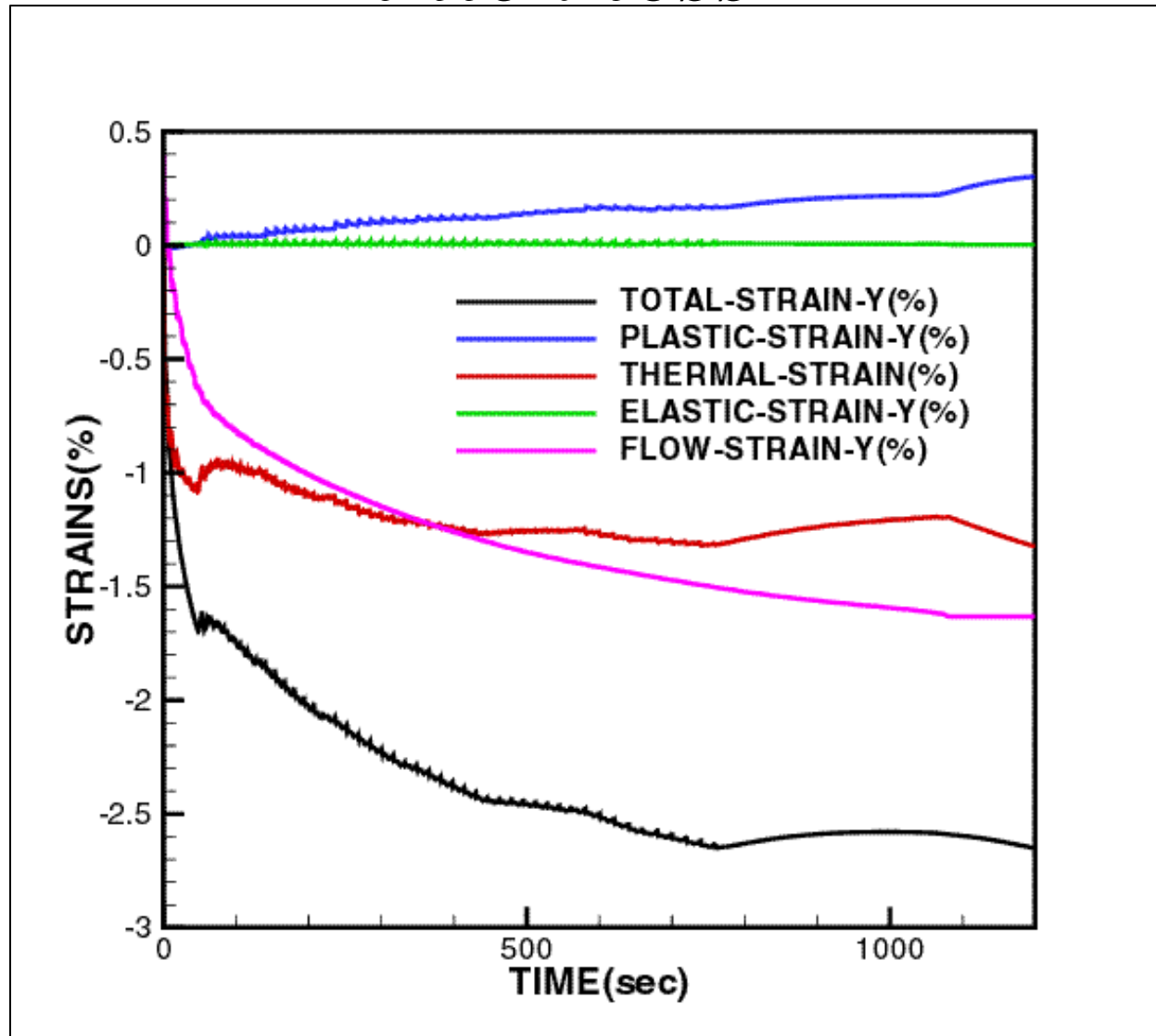
# *Normal force and shear stress applied on the domain*



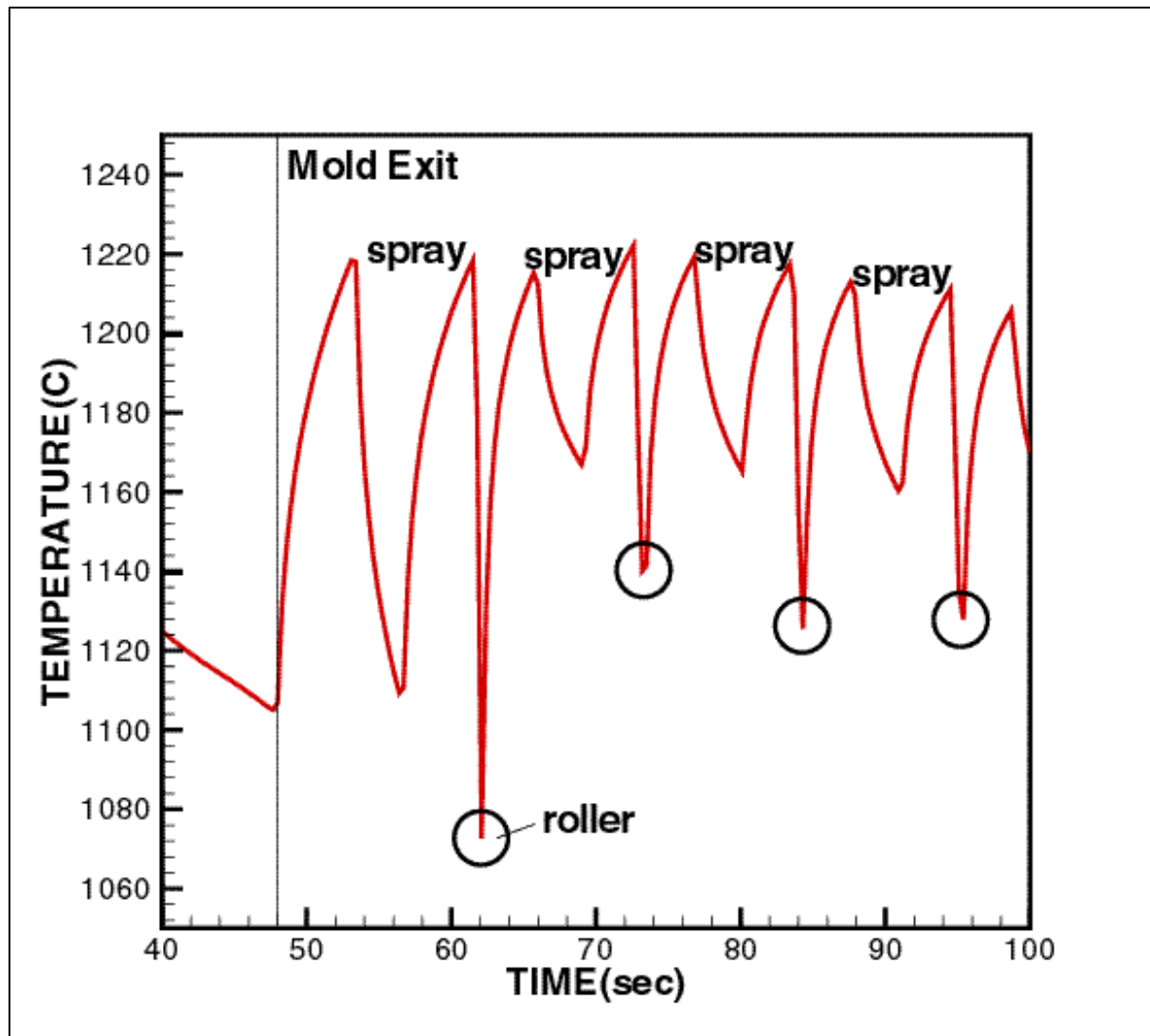
# *Average stress across shell thickness*



# *Average strains across shell thickness*

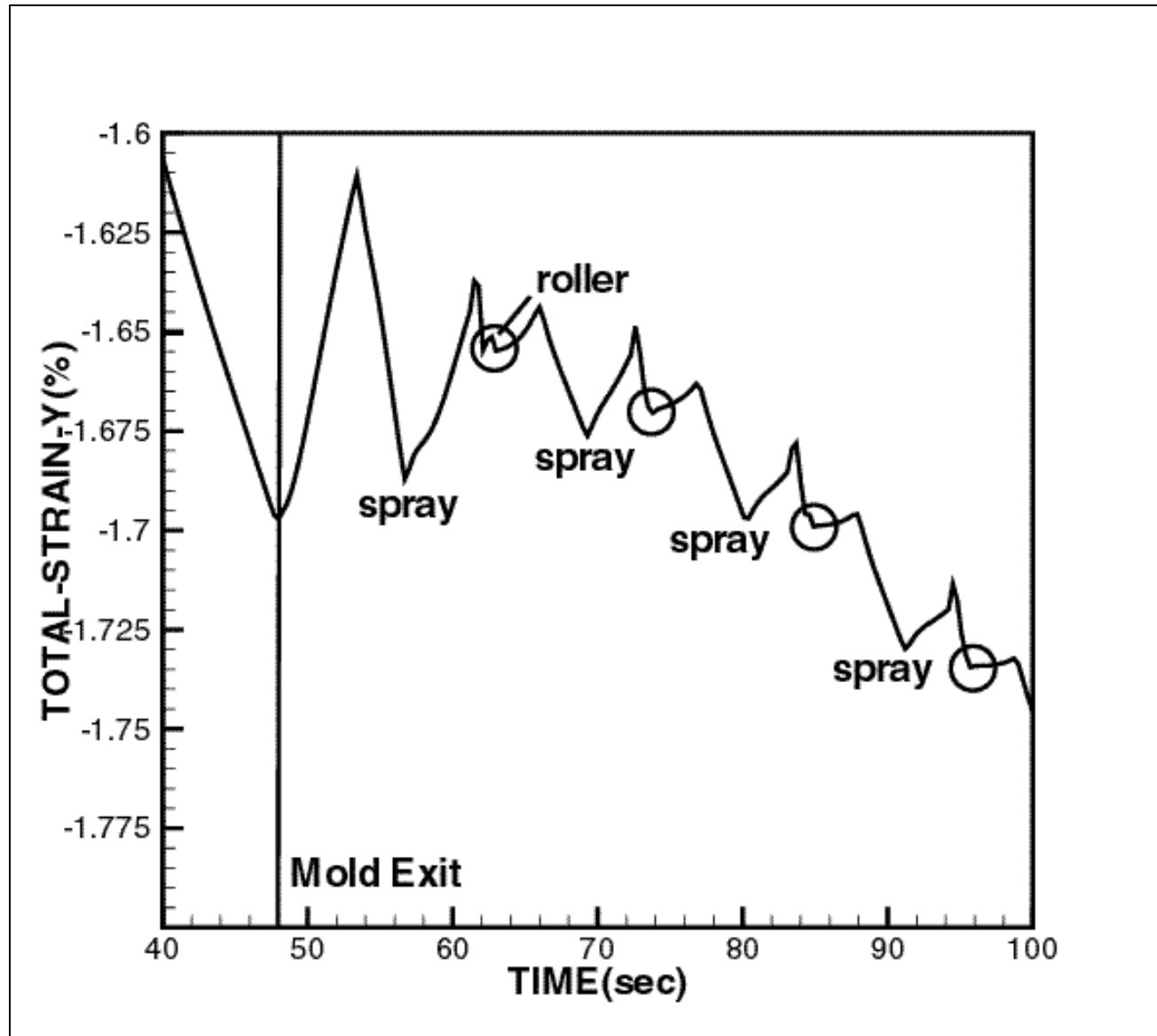


# *Surface temperature under first four rollers*

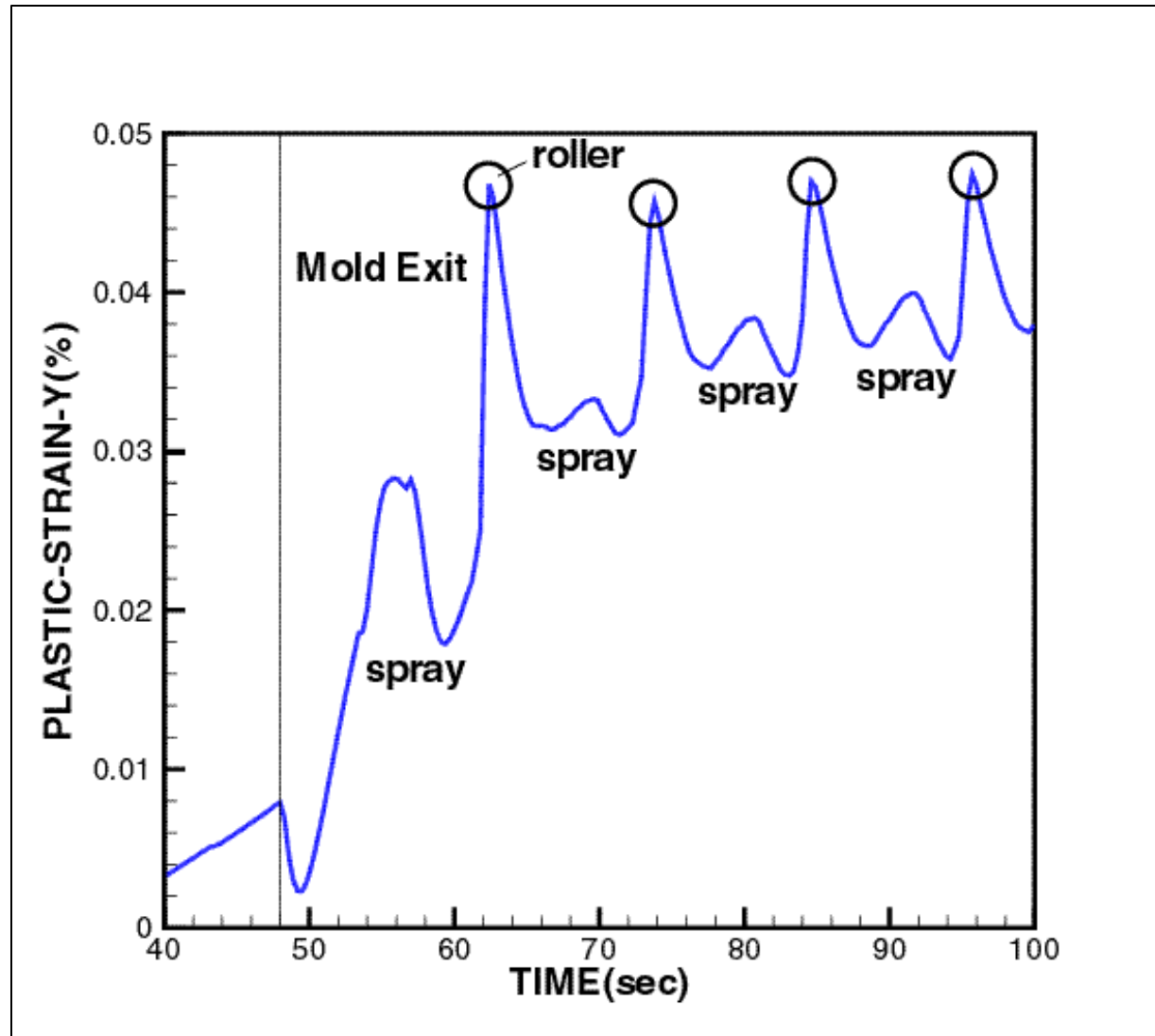




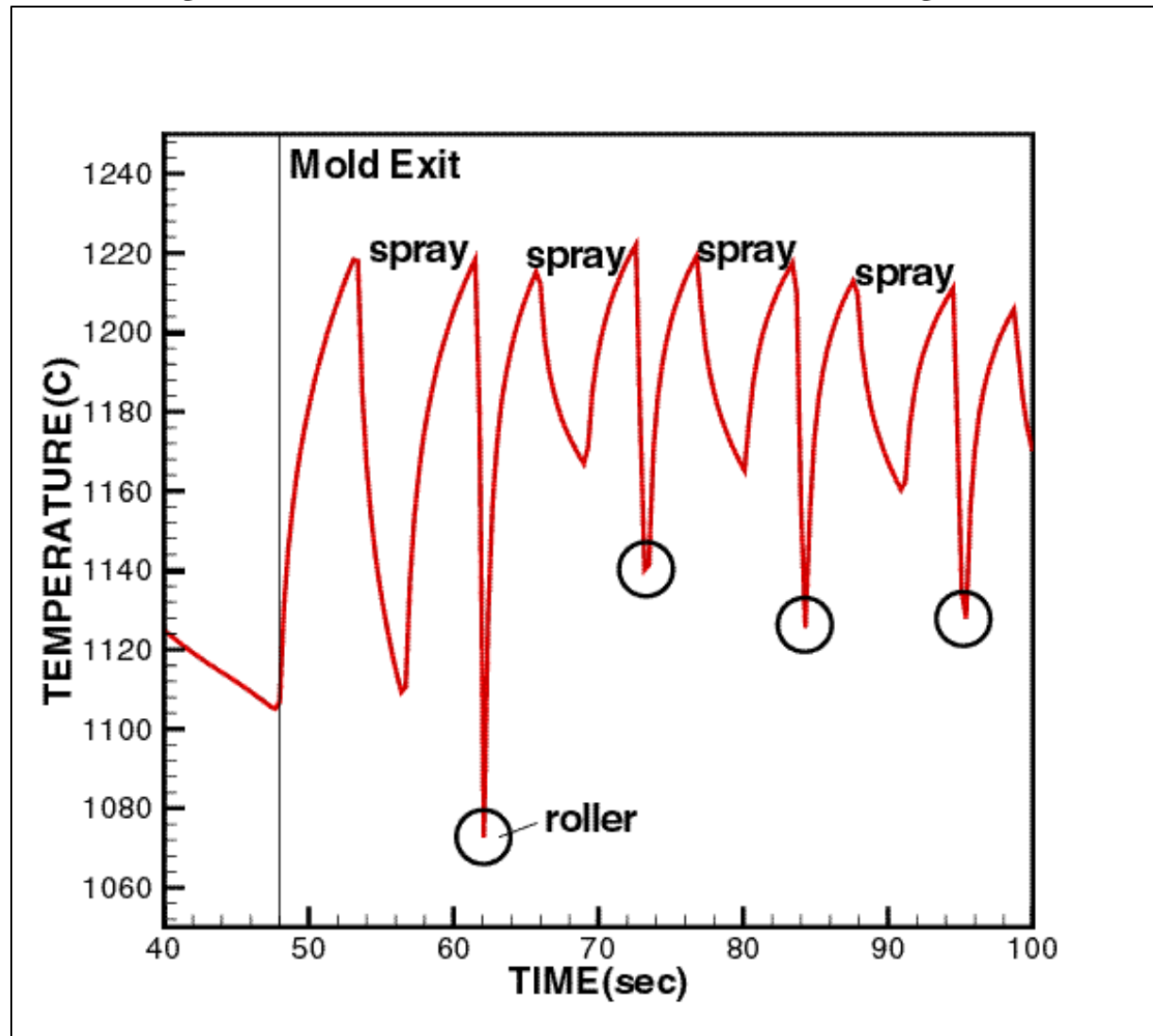
# *Total strain under first four rollers*



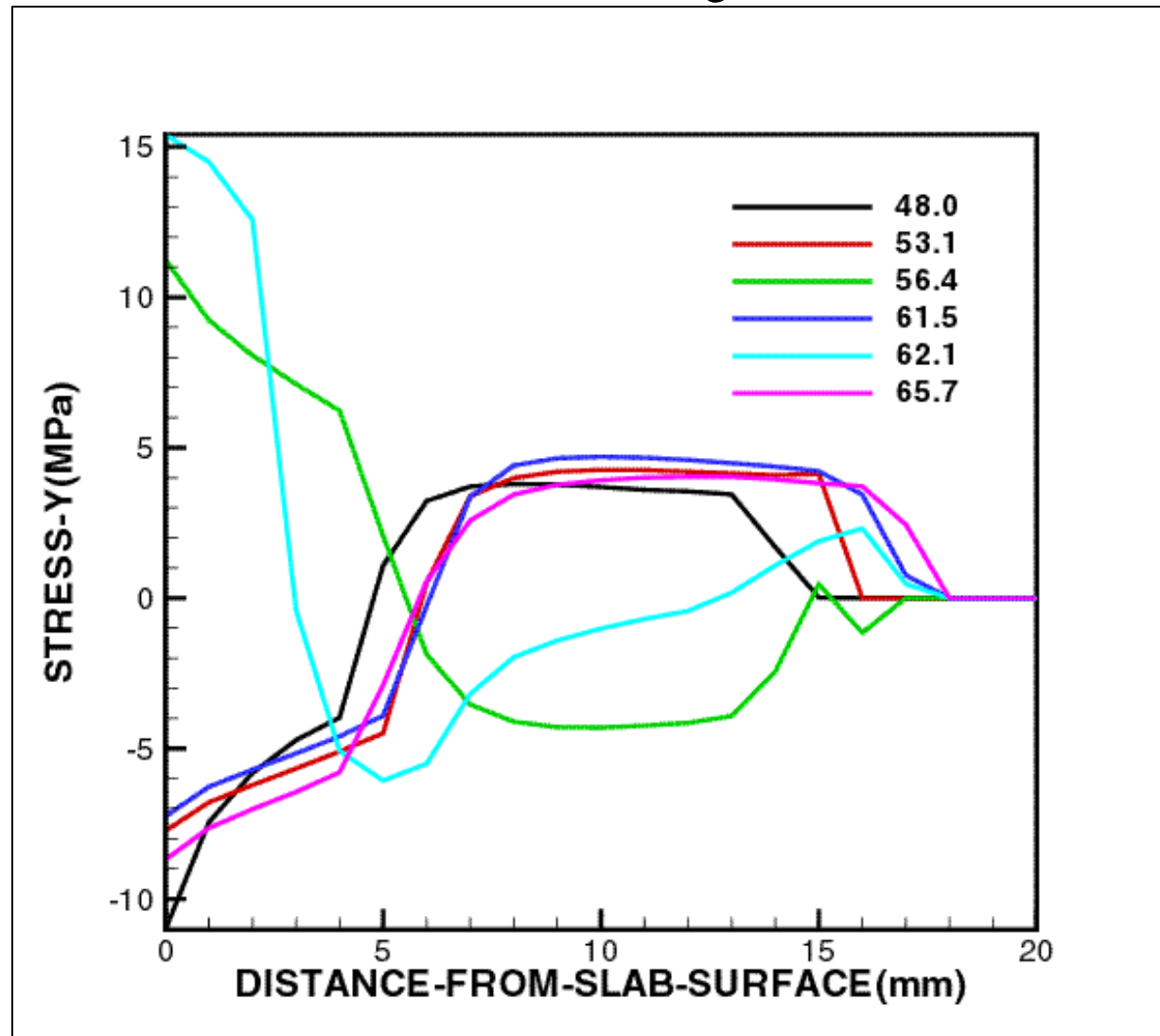
# *Average plastic strain under first four rollers*



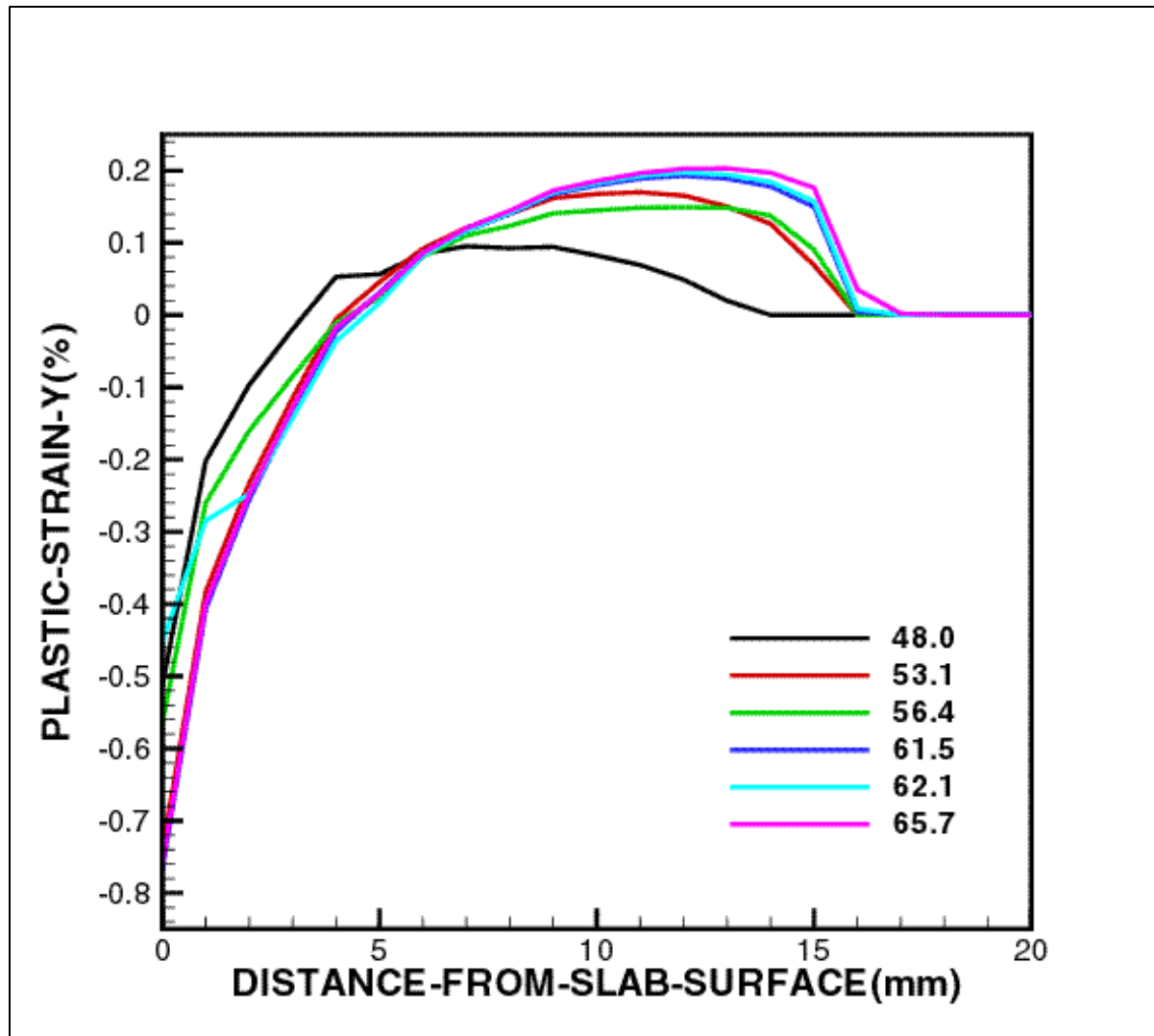
# *Temperatures through the shell thickness from mold exit to first roller*



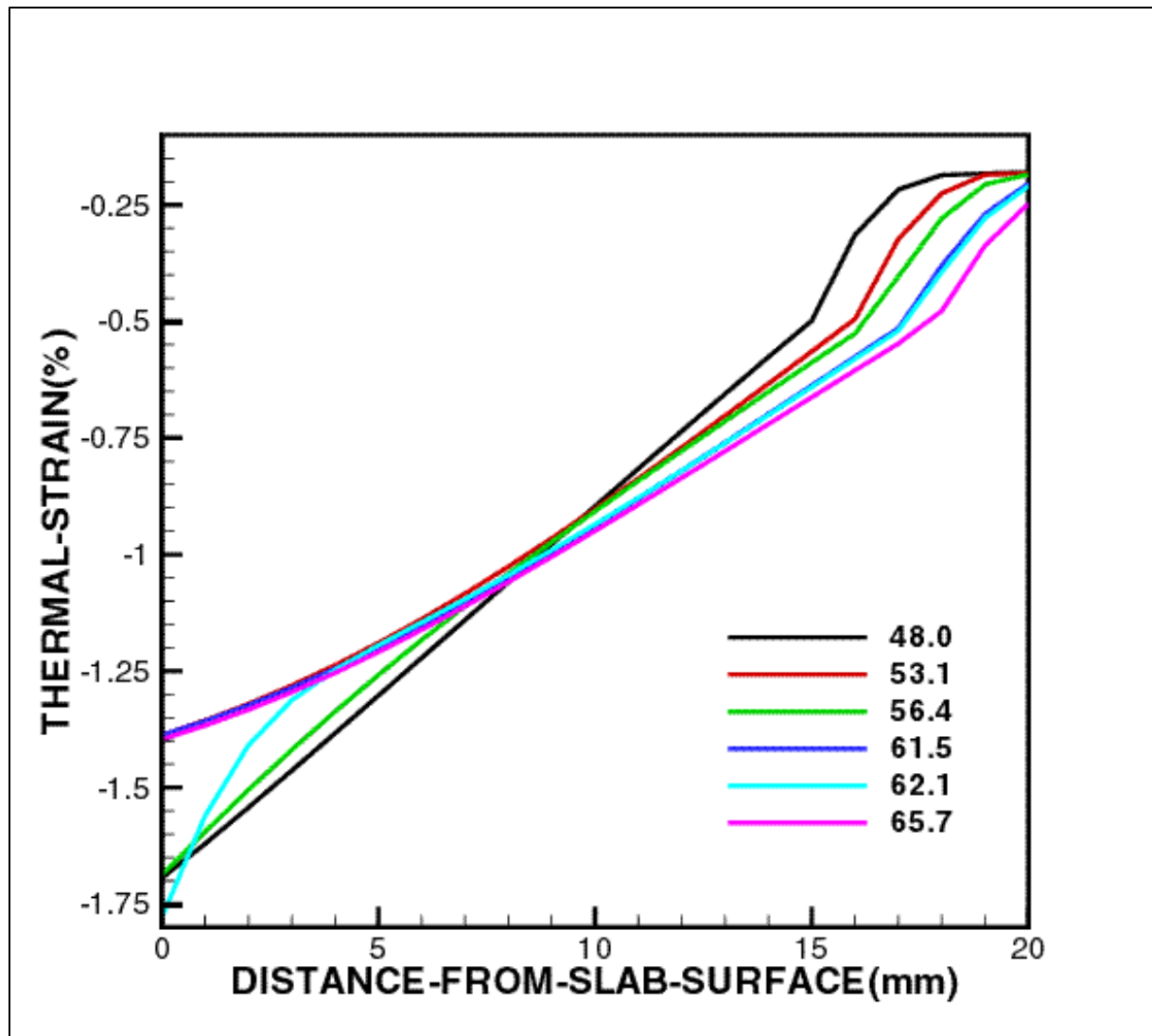
# *Stresses through the shell thickness from mold exit to first roller*



# *Plastic strain through shell thickness from mold exit to first roller*



# *Thermal strain through the shell thickness from mold exit to first roller*



# *Conclusions*

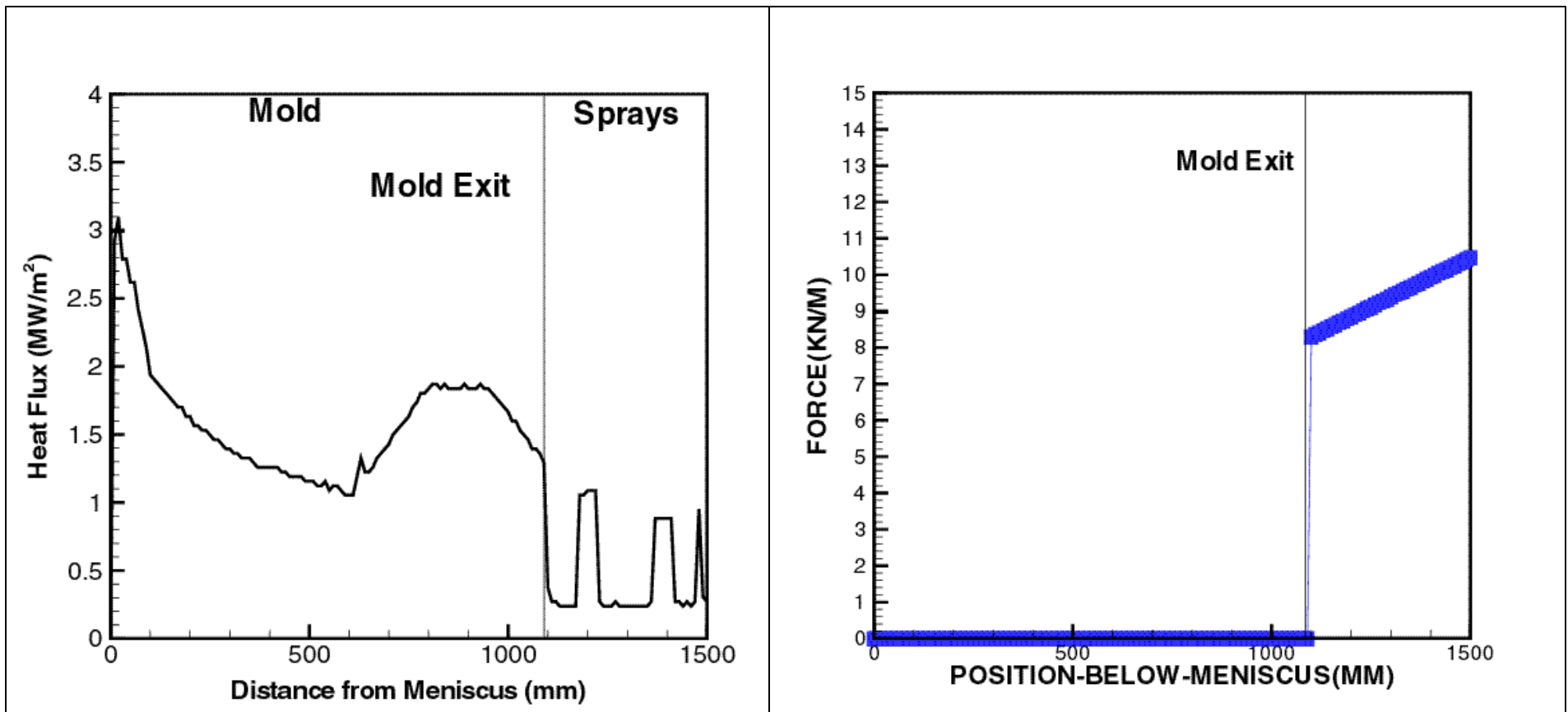
- *Ferrostatic pressure and friction force did make the shell expand through a ratcheting mechanism, however, they can not overcome the thermal shrinkage from cooling.*
- *Most of the load is taken by the outer layer of the solidifying shell less than 10 mm from the slab surface.*
- *Other mechanisms are needed to explain the width expansion phenomenon.*

## ***Future Work***

- *Extend the taper project to stainless steel.*
- *Link critical shell thickness results with lubrication predictions.*
- *Link critical shell thickness results with cracking criterion and new micro-segregation model.*
- *Pursuing new mechanisms to complete the strand width variation problem.*



# *Heat Flux and Force Applied on Shell Histories for Normal Case*



# *Heat Flux and Force Applied on Shell Histories for Critical Case*

